

NIST Interatomic Potentials Repository Project

Lucas M. Hale

NIST
National Institute of
Standards and Technology
U.S. Department of Commerce

IPR

**MATERIAL
MEASUREMENT
LABORATORY**

INTERATOMIC POTENTIALS REPOSITORY

<https://www.ctcms.nist.gov/potentials/>

260+ potentials

- Known provenance
- Any format
- Full citation and abstracts
- Property calculations

Incorporated into many projects

- OpenKIM
- JARVIS-FF
- pyiron
- MedeA

DOIs for hosted files in progress

2006--Williams-P-L-Mishin-Y-Hamilton-J-C--Ag

Citation: P.L. Williams, Y. Mishin, and J.C. Hamilton (2006), "An embedded-atom potential for the Cu-Ag system", *Modelling and Simulation in Materials Science and Engineering*, **14**(5), 817-833. DOI: [10.1088/0965-0393/14/5/002](https://doi.org/10.1088/0965-0393/14/5/002).

Abstract: A new embedded-atom method (EAM) potential has been constructed for Ag by fitting to experimental and first-principles data. The potential accurately reproduces the lattice parameter, cohesive energy, elastic constants, phonon frequencies, thermal expansion, lattice-defect energies, as well as energies of alternate structures of Ag. Combining this potential with an existing EAM potential for Cu, a binary potential set for the Cu-Ag system has been constructed by fitting the cross-interaction function to first-principles energies of imaginary Cu-Ag compounds. Although properties used in the fit refer to the 0 K temperature (except for thermal expansion factors of pure Cu and Ag) and do not include liquid configurations, the potentials demonstrate good transferability to high-temperature properties. In particular, the entire Cu-Ag phase diagram calculated with the new potentials in conjunction with Monte Carlo simulations is in satisfactory agreement with experiment. This agreement suggests that EAM potentials accurately fit to 0 K properties can be capable of correctly predicting simple phase diagrams. Possible applications of the new potential set are outlined.

EAM tabulated functions

Notes: These files were provided by Yuri Mishin.

File(s):

Ag F(ρ): [F_ag.plt](#)

Ag ρ (r): [fag.plt](#)

Ag ϕ (r): [pag.plt](#)

LAMMPS pair_style eam/alloy (2006--Williams-P-L--Ag--LAMMPS--ipr1)

[See Computed Properties](#)

Notes: This conversion was produced by Chandler Becker on 4 February 2009 from the plt files listed above. This version is compatible with LAMMPS. Validation and usage information can be found in [Ag06_releaseNotes_1.pdf](#). If you use this setfile, please credit the website in addition to the original reference.

File(s):

[Ag.eam.alloy](#)

[Ag06_releaseNotes_1.pdf](#)

INTERATOMIc POTENTIALS REPOSITORY

Show computed properties for comparing potentials
Full method descriptions + notes, disclaimers, version info
Interactive plots and downloadable raw data

Cohesive Energy vs. Interatomic Spacing

Plots of the cohesive energy vs interatomic spacing, r , are shown below for a number of crystal structures. The values were computed using the iprPy $E_{vs_r_scan}$ calculation method. The structures are generated based on the ideal atomic positions and b/a and c/a lattice parameter ratios for a given crystal prototype. The size of the system is then uniformly scaled, and the energy calculated without relaxing the system. To obtain these plots, values of r are evaluated every 0.02 Å up to 6 Å. Clicking on the image of a plot will open an interactive version of it in a new tab. The underlying data for the plots can be downloaded by clicking on the links above each plot.

Notes and Disclaimers:

- These values are meant to be guidelines for comparing potentials, not the absolute values for any potential's properties. Values listed here may change if the calculation methods are updated due to improvements/corrections. Variations in the values may occur for variations in calculation methods, simulation software and implementations of the interatomic potentials.
- The minima identified by this calculation do not guarantee that the associated crystal structures will be stable since no relaxation is performed.
- NIST disclaimer

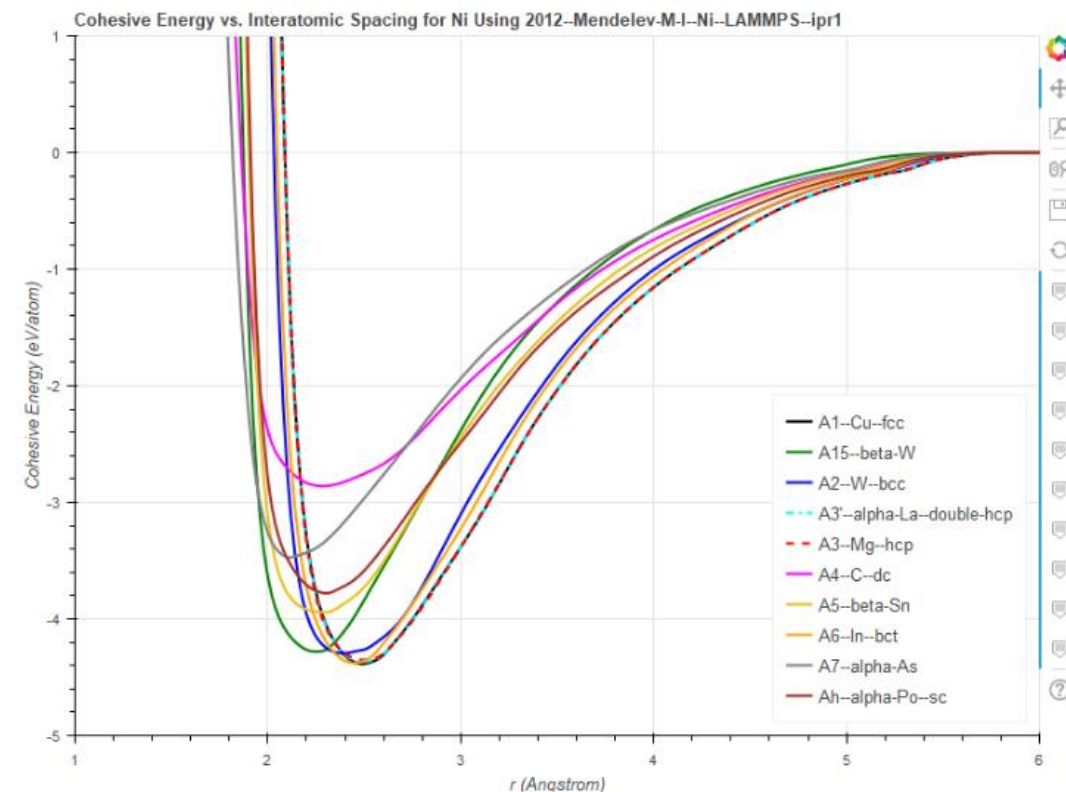
Version Information:

- 2019-02-04. Values regenerated with even r spacings of 0.02 Å, and now include values less than 2 Å when possible. Updated calculation method and parameters enhance compatibility with more potential styles.
- 2019-04-26. Results for hcp, double hcp, α -As and $L1_2$ prototypes regenerated from different unit cell representations. Only α -As results show noticeable ($>1e-5$ eV) difference due to using a different coordinate for Wyckoff site c position.
- 2018-06-13. Values for MEAM potentials corrected. Dynamic versions of the plots moved to separate pages to improve page loading. Cosmetic changes to how data is shown and updates to the documentation.
- 2017-01-11. Replaced png pictures with interactive Bokeh plots. Data regenerated with 200 values of r instead of 300.
- 2016-09-28. Plots for binary structures added. Data and plots for elemental structures regenerated. Data values match the values of the previous version. Data table formatting slightly changed to increase precision and ensure spaces between large values. Composition added to plot title and structure names made longer.
- 2016-04-07. Plots for elemental structures added.

Select a composition:

[Download data](#)

[Click on plot to load interactive version](#)



NEW CALCULATIONS

Elastic Constants Predictions

Static elastic constants are displayed for the unique structures identified in Crystal Structure Predictions above. The values displayed here are obtained by measuring the change in virial stresses due to applying small strains to the relaxed crystals. The initial structure and the strained states are all relaxed using force minimization.

Notes and Disclaimers:

- These values are meant to be guidelines for comparing potentials, not the absolute values for any potential's properties. Values listed here may change if the calculation methods are updated due to improvements/corrections. Variations in the values may occur for variations in calculation methods, simulation software and implementations of the interatomic potentials.
- The presence of any structures in this list does not guarantee that those structures are stable.
- The elastic constants have been computed for a variety of strains, and in some cases for slightly different lattice constant values. The static nature of this calculation can give poor predictions if the evaluated states straddle a functional discontinuity in the potential's third derivative. Be sure to compare the elastic constants for the different strains (positive and negative).
- [NIST disclaimer](#)

Version Information:

- 2019-08-07. Data added.

Composition:
 Prototype:
 a_0 :
 strain:
[Download raw data](#)

Cij in GPa:

124.239	93.874	93.874	-0.0	-0.0	0.0
93.874	124.239	93.874	-0.0	-0.0	0.0
93.874	93.874	124.239	-0.0	-0.0	0.0
0.0	0.0	0.0	46.419	0.0	-0.0
0.0	0.0	0.0	-0.0	46.419	0.0
0.0	0.0	0.0	-0.0	-0.0	46.419

Diatom Energy vs. Interatomic Spacing

Plots of the potential energy vs interatomic spacing, r , are shown below for all diatom sets associated with the interatomic potential. This calculation provides insights into the functional form of the potential's two-body interactions. The values were computed using the iprPy diatom_scan calculation method. A system consisting of only two atoms is created, and the potential energy is evaluated for the atoms separated by $0.02 \text{ \AA} \leq r \leq 6.0 \text{ \AA}$ in intervals of 0.02 \AA . Two plots are shown: one for the "standard" interaction distance range, and one for small values of r . The small r plot is useful for determining if the potential is suitable for radiation studies.

Clicking on the image of a plot will open an interactive version of it in a new tab. The underlying data for the plots can be downloaded by clicking on the links above each plot.

Notes and Disclaimers:

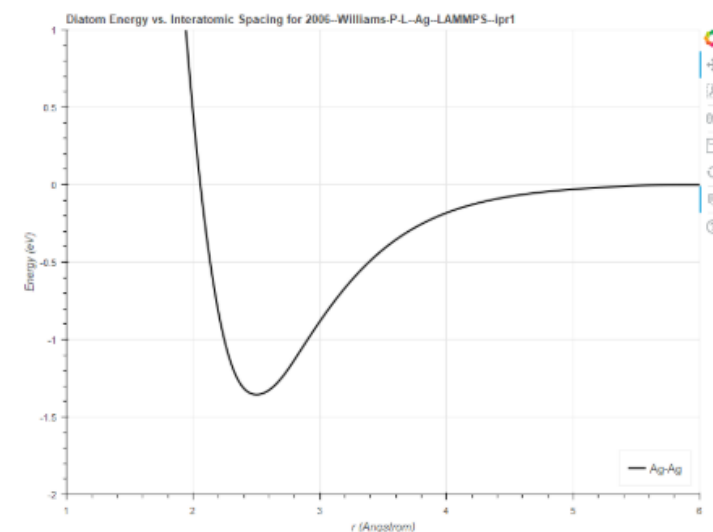
- These values are meant to be guidelines for comparing potentials, not the absolute values for any potential's properties. Values listed here may change if the calculation methods are updated due to improvements/corrections. Variations in the values may occur for variations in calculation methods, simulation software and implementations of the interatomic potentials.
- As this calculation only involves two atoms, it neglects any multi-body interactions that may be important in molecules, liquids and crystals.
- [NIST disclaimer](#)

Version Information:

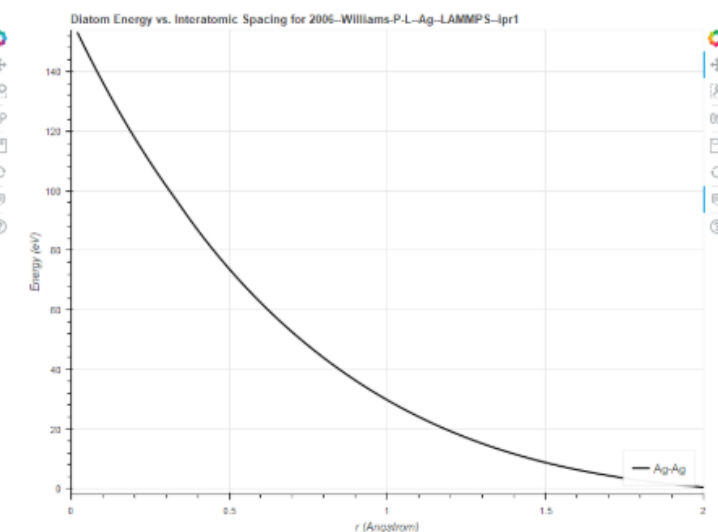
- 2019-08-07. Plots added.

[Download data](#)

Click on plot to load interactive version



Click on plot to load interactive version



NEW CALCULATIONS

Free Surface Formation Energy Predictions

Static free surface formation energies are displayed for select crystals. The values displayed here are obtained by taking a perfect periodic bulk crystal, slicing along a crystallographic plane, and using force minimization to statically relax the surfaces. The free surface formation energy is computed by comparing the energy of the defect system to the bulk system and dividing by the total surface area created by the cut.

Notes and Disclaimers:

- These values are meant to be guidelines for comparing potentials, not the absolute values for any potential's properties. Values listed here may change if the calculation methods are updated due to improvements/corrections. Variations in the values may occur for variations in calculation methods, simulation software and implementations of the interatomic potentials.
- The calculation only performs straight cuts along crystallographic planes and static relaxations. Lower energy configurations may exist that require atomic restructuring of the surfaces.
- Multiple values may be listed for a given plane followed by a number indicating different unique atomic planar cuts for the same theoretical plane. *NOTE: currently, all #2 variations need to be redone as the plane positions were accidentally left the same as the #1 variations.*
- [NIST disclaimer](#)

Version Information:

- 2019-08-07. Data added.

Composition:

Prototype:

a_0 :

[Download raw data](#)

Surface	γ_{fs} (mJ/m ²)
(331)	1002.7
(311)	1015.17
(110)	1016.82
(321)	1033.17
(310)	1046.67
(320)	1056.24
(210)	1061.75
(111)	862.21
(100)	940.43
(332)	942.78
(322)	950.37
(221)	976.61
(211)	990.88

Stacking Fault Energy Predictions

Stacking fault energy plots and maps are displayed for select crystals. The values are computed by

1. Starting with a bulk crystal system
2. Creating a free surface along one of the system's periodic boundaries and using force minimization to relax it
3. The system is sliced in half along a crystallographic plane parallel to the free surface. One half of the system is shifted relative to the other
4. The atoms in the shifted system are allowed to relax only in the direction normal to the shifting plane
5. The stacking fault energy for a given shift is computed by comparing the energy of the system before and after applying the shift, and dividing by the area of the fault plane

Notes and Disclaimers:

- These values are meant to be guidelines for comparing potentials, not the absolute values for any potential's properties. Values listed here may change if the calculation methods are updated due to improvements/corrections. Variations in the values may occur for variations in calculation methods, simulation software and implementations of the interatomic potentials.
- Values between the measured points are interpolated and therefore may not perfectly capture minima and maxima.
- Multiple values may be listed for a given plane followed by a number indicating different unique atomic planar cuts for the same theoretical plane. *NOTE: currently, all #2 variations need to be redone as the plane positions were accidentally left the same as the #1 variations.*
- [NIST disclaimer](#)

Version Information:

- 2019-08-07. Plots added.

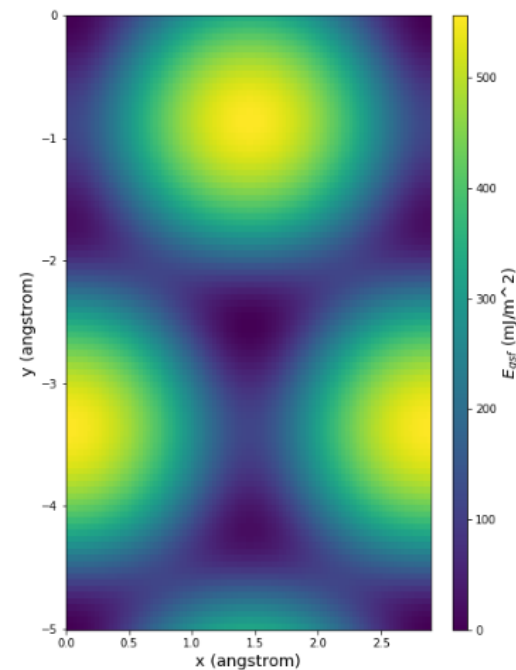
Composition:

Prototype:

a_0 :

plot:

[Download raw data](#)



NEW CALCULATIONS

Free Surface Formation Energy Predictions

Static free surface formation energies are displayed for select crystals. The values displayed here are obtained by taking a perfect periodic bulk crystal, slicing along a crystallographic plane, and using force minimization to statically relax the surfaces. The free surface formation energy is computed by comparing the energy of the defect system to the bulk system and dividing by the total surface area created by the cut.

Notes and Disclaimers:

- These values are meant to be guidelines for comparing potentials, not the absolute values for any potential's properties. Values listed here may change if the calculation methods are updated due to improvements/corrections. Variations in the values may occur for variations in calculation methods, simulation software and implementations of the interatomic potentials.
- The calculation only performs straight cuts along crystallographic planes and static relaxations. Lower energy configurations may exist that require atomic restructuring of the surfaces.
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- Values between the measured points are interpolated and therefore may not perfectly capture minima and maxima.
- Multiple values may be listed for a given plane followed by a number indicating different unique atomic planar cuts for the same theoretical plane. *NOTE: currently, all #2 variations need to be redone as the plane positions were accidentally left the same as the #1 variations.*
- [NIST disclaimer](#)

Version Information:

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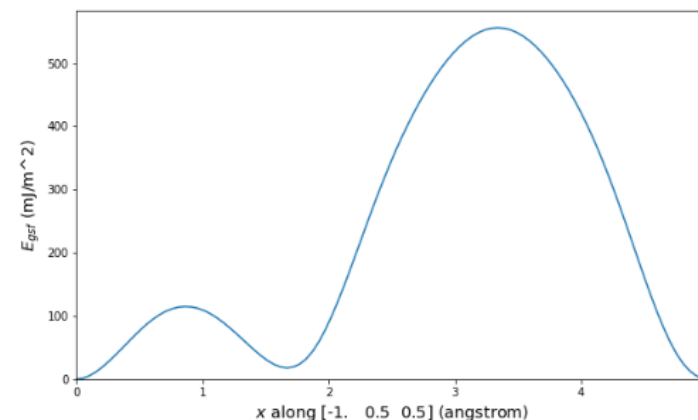
Composition:

Prototype:

a_0 :

plot:

[Download raw data](#)



NEW POTENTIALS DATABASE!

- <https://github.com/usnistgov/potentials>
- Add, edit, search, copy, share, don't share...
- Links to
- Early prototype, want feedback

Search for potentials

The Database.search_potentials() method currently allows for searches based on

- **author** (*str, optional*) Author string to search for. Note that the citation info must exactly contain this field, so multiple a
- **year** (*str, optional*) Publication year to search for.
- **elements** (*list, optional*) Element models to search for. If multiple elements are listed, the results will be inclusive, i.e. a elements will be included.

```
[ ] potentials = potdb.search_potentials(  
    #year = '2007',  
    author = 'Mendeleev',  
    elements = 'Fe',  
)  
  
# Show ids for all matching potentials  
for potential in potentials:  
    print(potential.id)
```

```
2003--Mendeleev-M-I-Han-S-Srolovitz-D-J-et-al--Fe-2  
2003--Mendeleev-M-I-Han-S-Srolovitz-D-J-et-al--Fe-5  
2004--Ackland-G-J-Mendeleev-M-I-Srolovitz-D-J-et-al--Fe-P  
2005--Mendeleev-M-I-Srolovitz-D-J-Ackland-G-J-Han-S--Al-Fe  
2007--Mendeleev-M-I-Han-S-Son-W-et-al--V-Fe
```

Show all implementations for a given potential

```
[ ] # Select a potential  
potential = potentials[0]  
  
# Display full HTML content for that potential  
display(HTML(potdb.full_html(potentials[0])))
```



2003--Mendeleev-M-I-Han-S-Srolovitz-D-J-et-al--Fe-2

Citation: M.I. Mendeleev, S. Han, D.J. Srolovitz, G.J. Ackland, D.Y. Sun and M. Asta (2003), "Development of new interatomic potentials for iron and nickel using the embedded-atom method (EAM) form and applied to determine the structure of liquid iron and nickel. The new potentials (provided herein) are, on average, in better agreement with the experimental or first-principles lattice parameters.

Notes: This listing is for the reference's Fe #2 interaction parameters.

LAMMPS pair_style eam/fs (2003--Mendeleev-M-I--Fe-2--LAMMPS--ipr1)
superseded

Notes: This file was provided by Mikhail Mendeleev on Jun 10, 2007. Except for comments, this file is identical to "Fe_mm.eam.fs" in the original repository.

Files:
[Fe_2_eam.fs](#)

LAMMPS pair_style eam (2003--Mendeleev-M-I--Fe-2--LAMMPS--ipr2)
retracted

Notes: Update 09 Mar 2009: The file for Fe #2 (Feb 22, 2009) was sent as a replacement for the Jun 10, 2007 file above. It better reproduces the experimental eV/atom for bcc with a=2.855324 Å. For archival purposes, the file can be found here. Thanks to Jianyang Wu for bringing this to our attention.

Files:
[Fe_2_eam](#)

LAMMPS pair_style eam/fs (2003--Mendeleev-M-I--Fe-2--LAMMPS--ipr3)

Notes: This file supports radial distances smaller than 0.5 Å and gives the proper values of -4.1224351 eV/atom for a = 2.855324 Å (Lennard-Jones parameters).

Files:
[Fe_2_eam.fs](#)

OpenKIM (MO_769582363439)

Notes: Taken from <https://openkim.org>. This KIM potential is based on the files from 2003--Mendeleev-M-I--Fe-2--LAMMPS--ipr3.

Links:
[EAM_Dynamo_MendeleevHanSrolovitz_2003Potential2_Fe_MO_769582363439_005](#)

OpenKIM (MO_856295952425)

Notes: Taken from <https://openkim.org>.

Links:
[EAM_MagneticCubic_MendeleevHanSrolovitz_2003_Fe_MO_856295952425_002](#)

OpenKIM (MO_546673549085)

Notes: Taken from <https://openkim.org>. This KIM potential is based on the files from 2003--Mendeleev-M-I--Fe-2--LAMMPS--ipr3.

Links:
[EAM_Dynamo_Mendeleev_2003_Fe_MO_546673549085_000](#)

OpenKIM (MO_807997826449)

superseded

Notes: Taken from <https://openkim.org>. This KIM potential is based on the files from 2003--Mendeleev-M-I--Fe-2--LAMMPS--ipr1.

Links:
[EAM_Dynamo_MendeleevHanSrolovitz_2003_Fe_MO_807997826449_000](#)

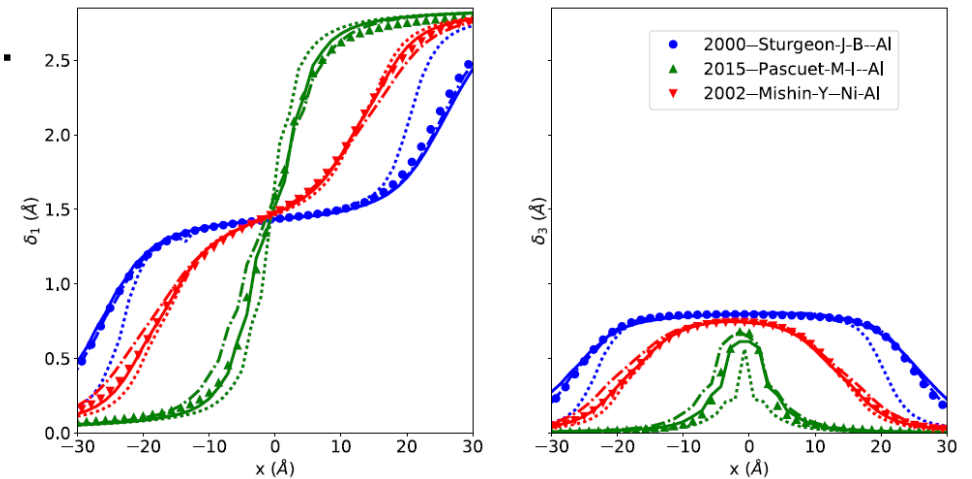
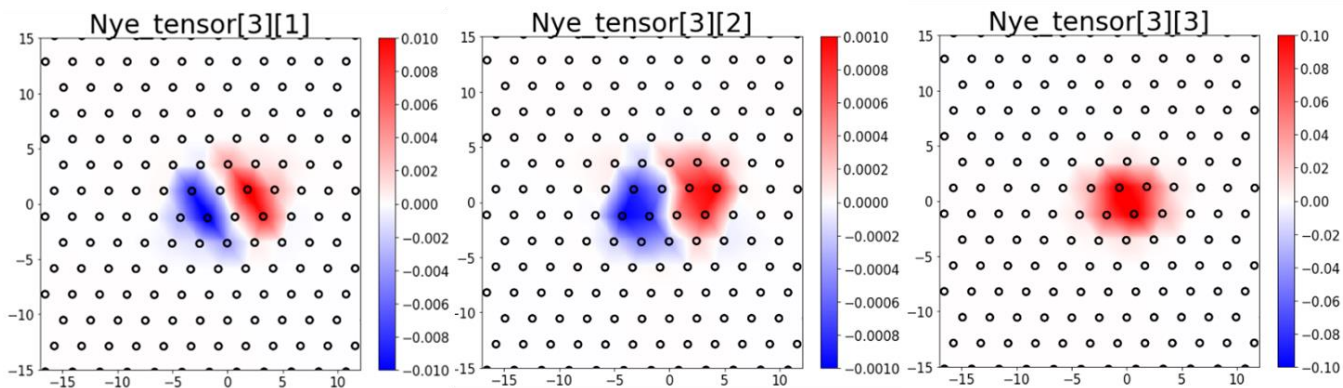
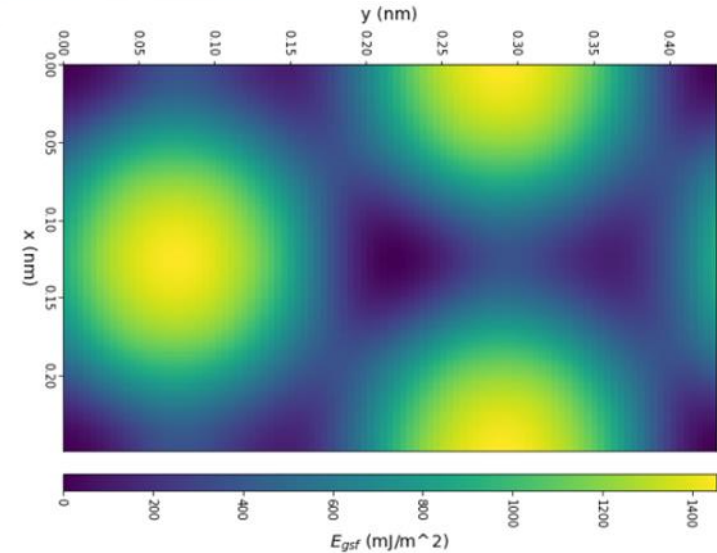
ATOMMAN : ATOMISTIC MANIPULATION TOOLKIT

<https://www.ctcms.nist.gov/potentials/atomman>

```
pip install atomman
```

Generic atomic representation designed to support large-scale MD

- Focus on defect generation + analysis
- Potential and simulator agnostic
- Converters to/from ase, pymatgen, spglib, ...



IPRPY CALCULATION FRAMEWORK

High-throughput calculation methods

- source: <https://github.com/usnistgov/iprPy>
- docs: <https://www.ctcms.nist.gov/potentials/iprPy>

Make calculation methods as accessible as possible

- Openly available
- Low barrier for usage
- Transparent, documented methodologies
- Adaptable to new materials
- Transferable to other frameworks

IPRPY CALCULATION FRAMEWORK

Python script: pass parameter file to stand-alone script

Python class: call calculation methods directly

Jupyter Notebook: single document with documentation, code and example

High-throughput: prepare and execute with lists of parameter values

```
# Input script for calc_E_vs_r_scan.py

# Command lines for LAMMPS and MPI
lammps_command      lmp_serial
mpi_command

# Potential definition and directory containing associated files
potential_file      1989--Adams-J-B--Ag--LAMMPS--ipr1.json
potential_dir       1989--Adams-J-B--Ag--LAMMPS--ipr1

# Initial system configuration to load
load_file           Al--Cu--fcc.json
load_style          system_model
load_options
family
symbols            Ag
box_parameters

# System manipulations
a_uvw
b_uvw
c_uvw
atomshift
sizemults          5 5 5

# Units for input/output values
length_unit
pressure_unit
energy_unit
force_unit

# Run parameters
minimum_r           0.5
maximum_r           6
number_of_steps_r   276
```

IPRPY CALCULATION FRAMEWORK

Python script: pass parameter file to stand-alone script

Python class: call calculation methods directly

Jupyter Notebook: single document with documentation, code and example

High-throughput: prepare and execute with lists of parameter values

```
import atomman as am
import iprPy

cohesive_scan = iprPy.load_calculation('E_vs_r_scan')

lammers_command = 'lmp_mpi'

ucell = am.load('system_model', 'Al--Cu--fcc.json', symbols = 'Ag')
system = ucell.supersize(5, 5, 5)

potential = am.lammers.Potential('1989--Adams-J-B--Ag--LAMMPS--ipr1.json')

rmin = 0.5
rmax = 6
rsteps = 276

cohesive_scan.calc(lammers_command, system, potential, ucell, rmin, rmax, rsteps)
```

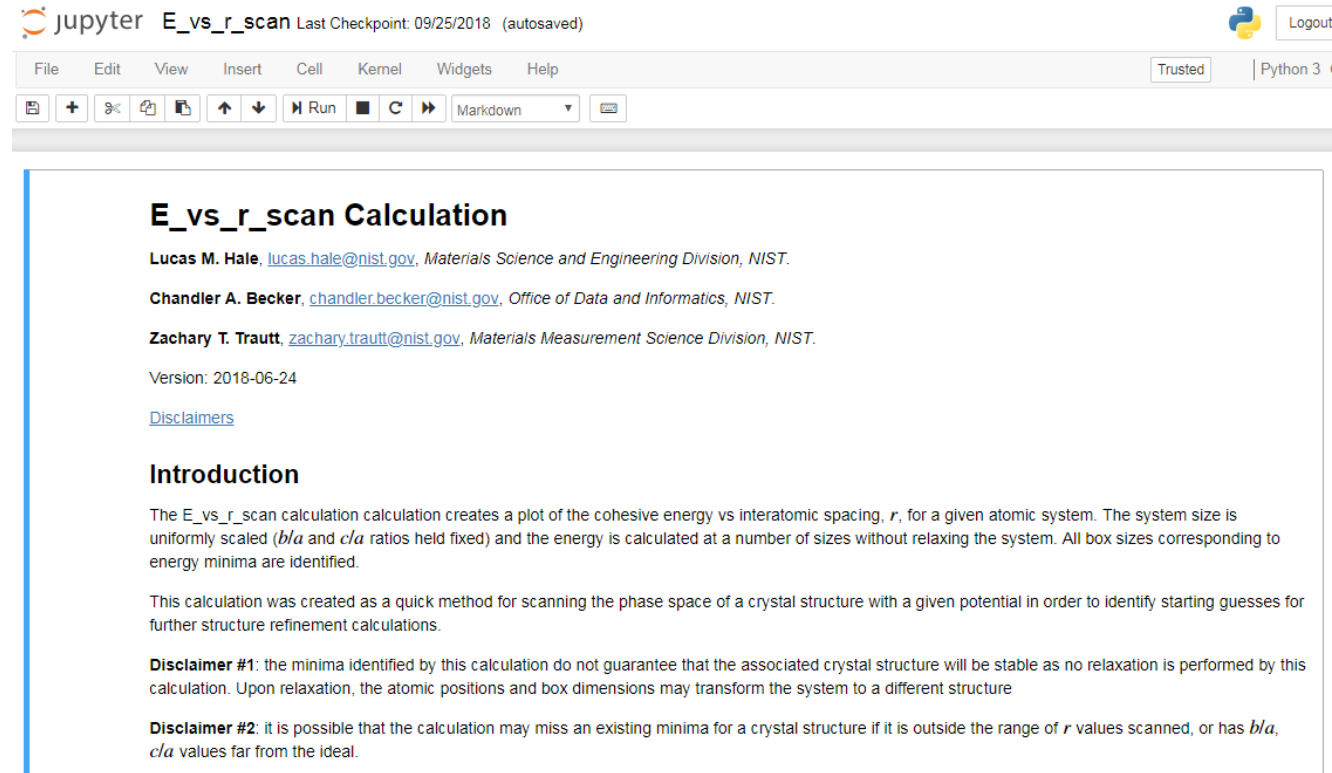
IPRPY CALCULATION FRAMEWORK

Python script: pass parameter file to stand-alone script

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jupyter E_vs_r_scan Last Checkpoint: 09/25/2018 (autosaved) Python 3

File Edit View Insert Cell Kernel Widgets Help Trusted

Run

E_vs_r_scan Calculation

Lucas M. Hale, lucas.hale@nist.gov, Materials Science and Engineering Division, NIST.

Chandler A. Becker, chandler.becker@nist.gov, Office of Data and Informatics, NIST.

Zachary T. Trautt, zachary.trautt@nist.gov, Materials Measurement Science Division, NIST.

Version: 2018-06-24

[Disclaimers](#)

Introduction

The E_vs_r_scan calculation creates a plot of the cohesive energy vs interatomic spacing, r , for a given atomic system. The system size is uniformly scaled (b/a and c/a ratios held fixed) and the energy is calculated at a number of sizes without relaxing the system. All box sizes corresponding to energy minima are identified.

This calculation was created as a quick method for scanning the phase space of a crystal structure with a given potential in order to identify starting guesses for further structure refinement calculations.

Disclaimer #1: the minima identified by this calculation do not guarantee that the associated crystal structure will be stable as no relaxation is performed by this calculation. Upon relaxation, the atomic positions and box dimensions may transform the system to a different structure

Disclaimer #2: it is possible that the calculation may miss an existing minima for a crystal structure if it is outside the range of r values scanned, or has b/a , c/a values far from the ideal.

Method and Theory

An initial system (and corresponding unit cell system) is supplied. The r/a ratio is identified from the unit cell. The system is then uniformly scaled to all r_i values in the range to be explored and the energy for each is evaluated using LAMMPS and "run 0" command, i.e. no relaxations are performed.

In identifying energy minima along the curve, only the explored values are used without interpolation. In this way, the possible energy minima structures are identified for r_i where $E(r_i) < E(r_{i-1})$ and $E(r_i) < E(r_{i+1})$.

Demonstration

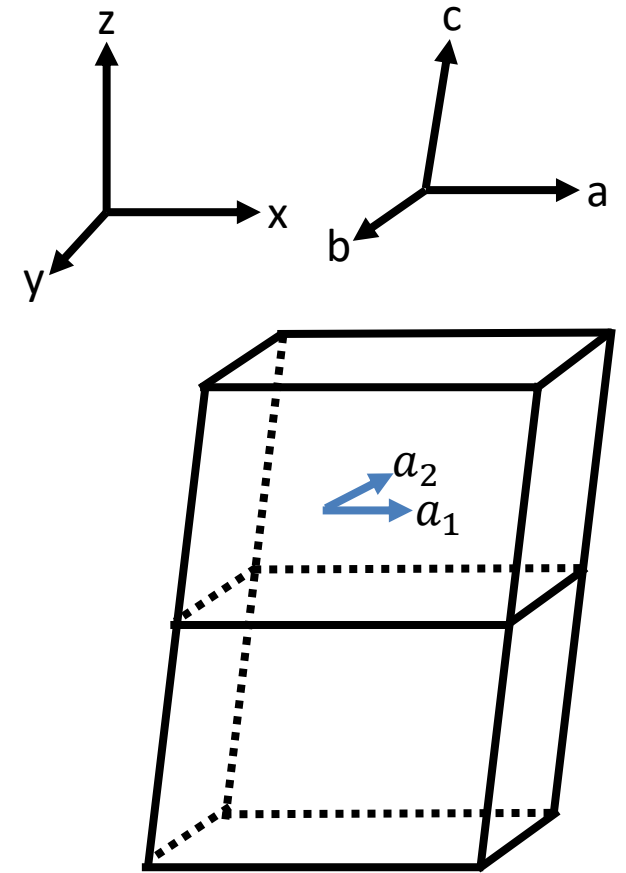
STACKING FAULT GENERATION AND PLOTTING

Method relatively simple

- 2 directions periodic, one not (free surface)
- Shift half of system by in-plane vectors a_1, a_2
- Measure $E(a_1, a_2)$

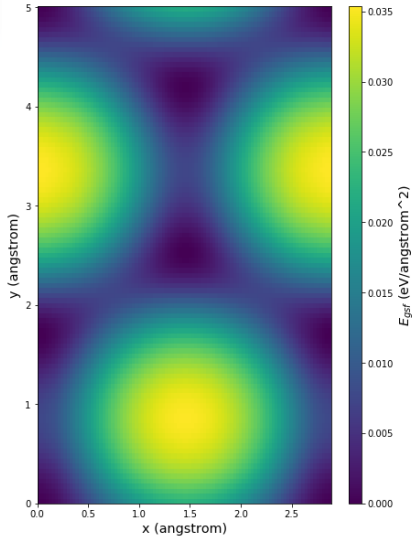
Tricky for general system

- Fault planes given as Miller crystallographic (hkl) planes and [uvw] vectors relative to conventional unit cells
- $[hkl] \perp (hkl)$ only guaranteed for cubic systems
- System box vectors rotated from unit cell box vectors
- System box vectors a, b, c not aligned with x, y, z
- a_1, a_2 may or may not be aligned with a, b

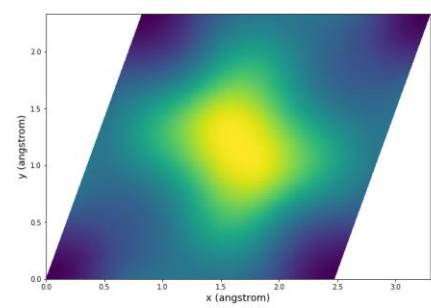


STACKING FAULT GENERATION AND PLOTTING

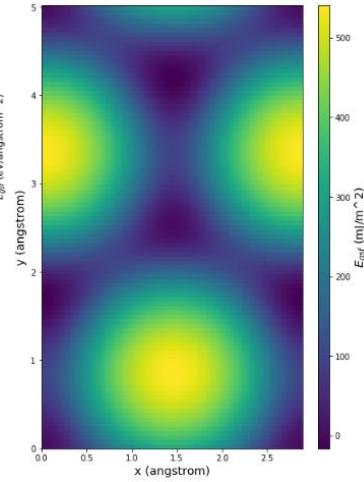
FCC (111)



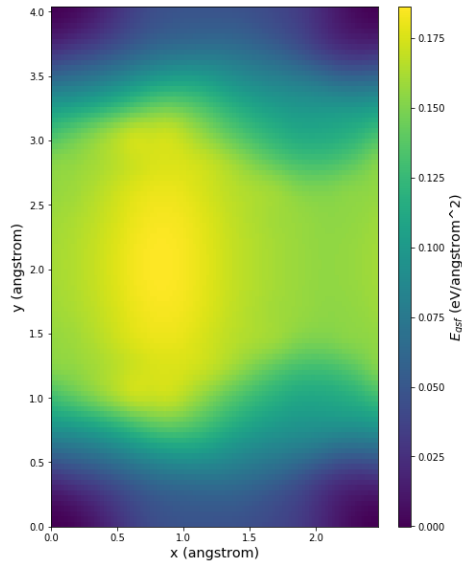
BCC (110)



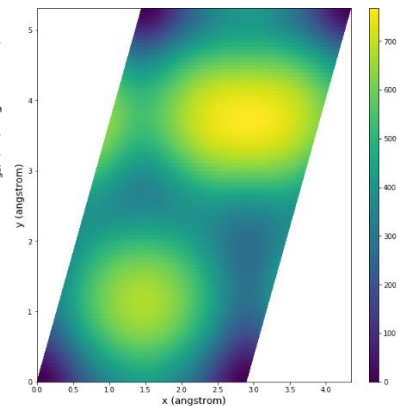
HCP (0001)



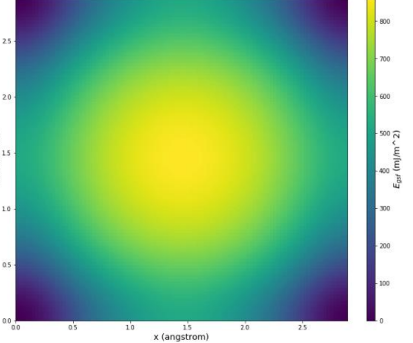
BCC (112)



HCP (1011)



FCC (100)



Initialize the StackingFault object and check parameters

```
[11]: sf = am.defect.StackingFault(system, a1vect=a1vect, a2vect=a2vect, ucellbox=ucell.box, transform=tra

print('Fault system has', sf.system.natoms, 'total atoms.')
print('Fault system has', len(sf.system.atoms.atype[sf.abovefault]), 'atoms above the fault')
print('Fault plane is at z =', sf.faultposcart)
print()
print("Fault system's pbc is", sf.system.pbc)
print()
print('a1vect as given:', sf.a1vect)
print('Cartesian a1vect:', sf.a1vectcart)
print()
print('a2vect as given:', sf.a2vect)
print('Cartesian a2vect:', sf.a2vectcart)
```

Fault system has 48 total atoms.
 Fault system has 24 atoms above the fault
 Fault plane is at z = 14.029611541307906

Fault system's pbc is [True True False]

a1vect as given: [-0.5 0.5 0.]
 Cartesian a1vect: [2.86378246e+00 -2.75990494e-16 -4.17652603e-16]

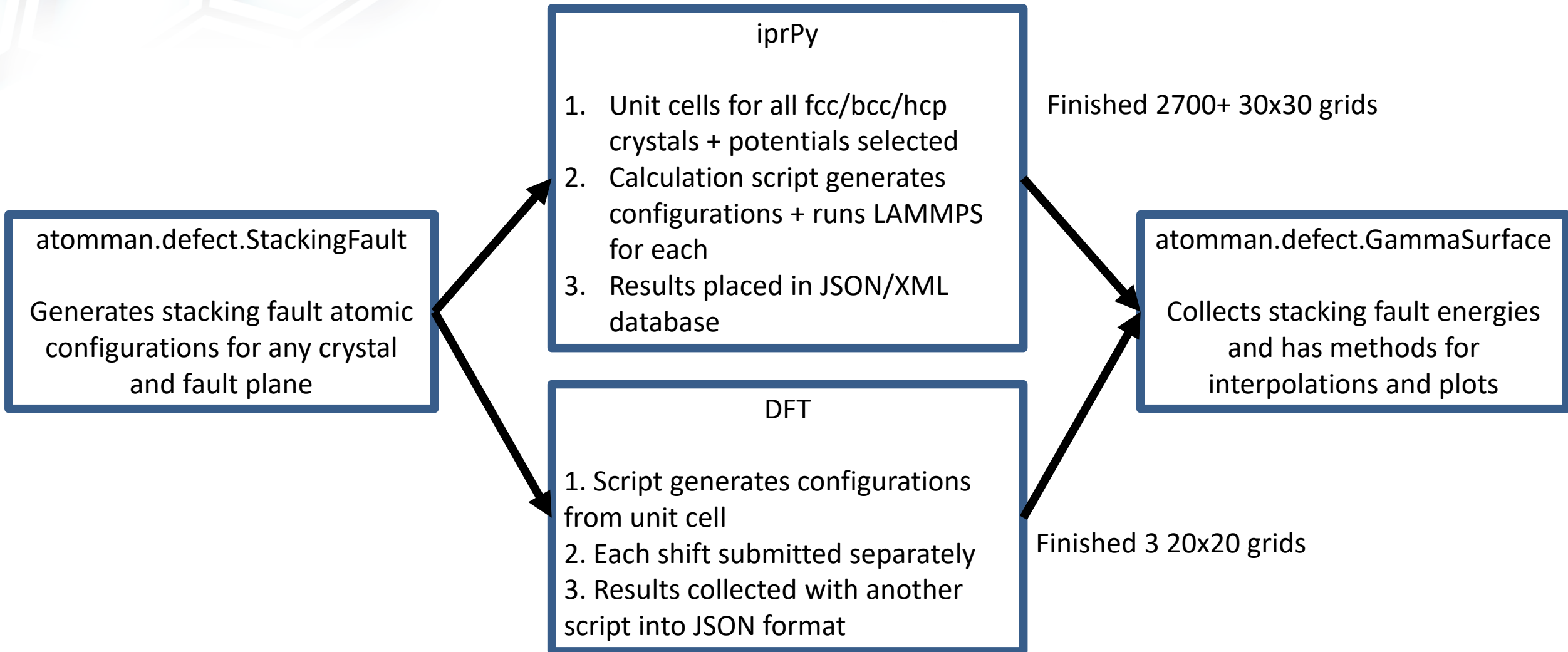
a2vect as given: [-0.5 0. 0.5]
 Cartesian a2vect: [1.43189123e+00 2.48010836e+00 -4.17652603e-16]

Now, StackingFault.iterfaultmap() can generate a 2D map. Note that the $\langle \bar{1}\bar{1}2 \rangle$ path corresponds to the points where $a_1 = a_2$

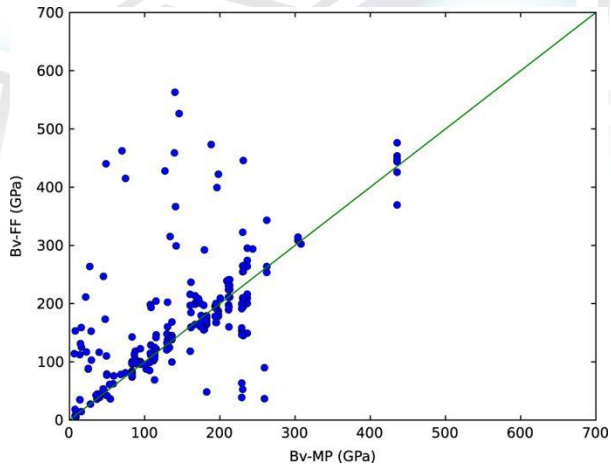
```
[12]: for a1, a2, faultsystem in sf.iterfaultmap(num_a1=5, num_a2=5):
    d = am.displacement(system, faultsystem)[-1]
    print('a1 = %.1f, a2 = %.1f, d = [%7.4f %7.4f %7.4f]' % (a1, a2, d[0], d[1], d[2]))
```

a1 = 0.0, a2 = 0.0, d = [0.0000 0.0000 0.0000]
 a1 = 0.2, a2 = 0.0, d = [0.5728 0.0000 0.0000]
 a1 = 0.4, a2 = 0.0, d = [1.1455 0.0000 0.0000]
 a1 = 0.6, a2 = 0.0, d = [1.7183 -0.0000 0.0000]
 a1 = 0.8, a2 = 0.0, d = [2.2910 -0.0000 0.0000]

HIGH-THROUGHPUT GAMMA SURFACES



JARVIS-FF



- Comparing classical and quantum data
 - Selecting appropriate potentials/FFs
 - Reliability of FFs
- Easy web-based search
- Automating atomistic calculations (all scripts on github)
- ~50000 LAMMPS calculations, ~1500 materials
- Machine learning

<https://jarvis.nist.gov/>

HOME JARVIS-DFT JARVIS-FF DOCUMENTATION OTHER

[NIST Disclaimer](#)

Structural formula: Al

Force-field: Mishin-Ni-Al-2009.eam.alloy

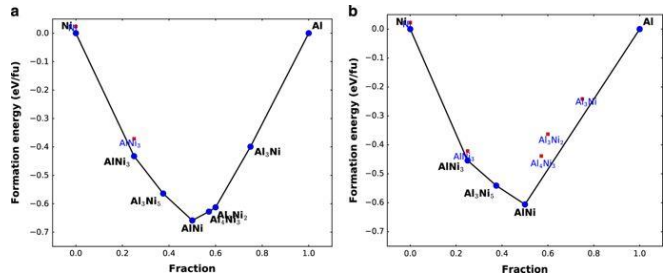
Space group : Fm-3m

JARVIS ID: JLMP-1243

[Download input files](#)

```

HM: P 1
a=12.150Å
b=12.150Å
c=12.150Å
α=90.000°
β=90.000°
γ=90.000°
    
```



Vacancy-formation energy (eV)

Vacancy formation energies were calculated by deleting the symmetrically distinct atoms in the system. The reference element cohesive energies were calculated with the most stable structure for the element as input. For defect-structures energetics calculations, constant volume ensemble was used.

Element	Mult.	Value
Al	4	0.674

[Download cif file](#)

Surface energy (J/m²)

Surface energies were calculated for symmetrically distinct surfaces. The slab thickness to be at least 2 nm and vacuum size to be at least 10 Å.

JARVIS-FF

Simple search:
Tell me about silicon-germanium / Si-Ge- / click Si, Ge, then hit enter or click Search

Advanced search:
JARVIS-ID: JLMP-167

H																	He															
Li	Be															B	C	N	O	F	Ne											
Na	Mg															Al	Si	P	S	Cl	Ar											
K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr															
Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xe															
Cs	Ba	*	Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn															
Fr	Ra	*	Rf	Db	Sg	Bh	Hs	Mt	Ds	Rg	Cn	Uut	Uuq	Uup	Uuh	Uus	Uuo															
																		La	Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb	Lu
																		Ac	Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No	Lr

Phonon

Phonons were obtained by making an interface of JARVIS-FF with Phonopy package at 0 K [Source]. For deformed-structures, constant volume ensemble was used. The band-indices for phonon bandstructure was obtained with Pymatgen. The phonon representation were obtained with Phonopy.

- Coming soon:**
- Grain-boundary energies,
 - Stacking faults,
 - Machine learning FFs

See also

Links to other databases or papers are provided below

[JVASP-816](#)

[mp-134](#)

Energy above hull from mp (eV): 0.0

MATERIALS RESOURCE REGISTRY

<https://materials.registry.nist.gov/>

“Yellow pages” for materials tools

Search current entries

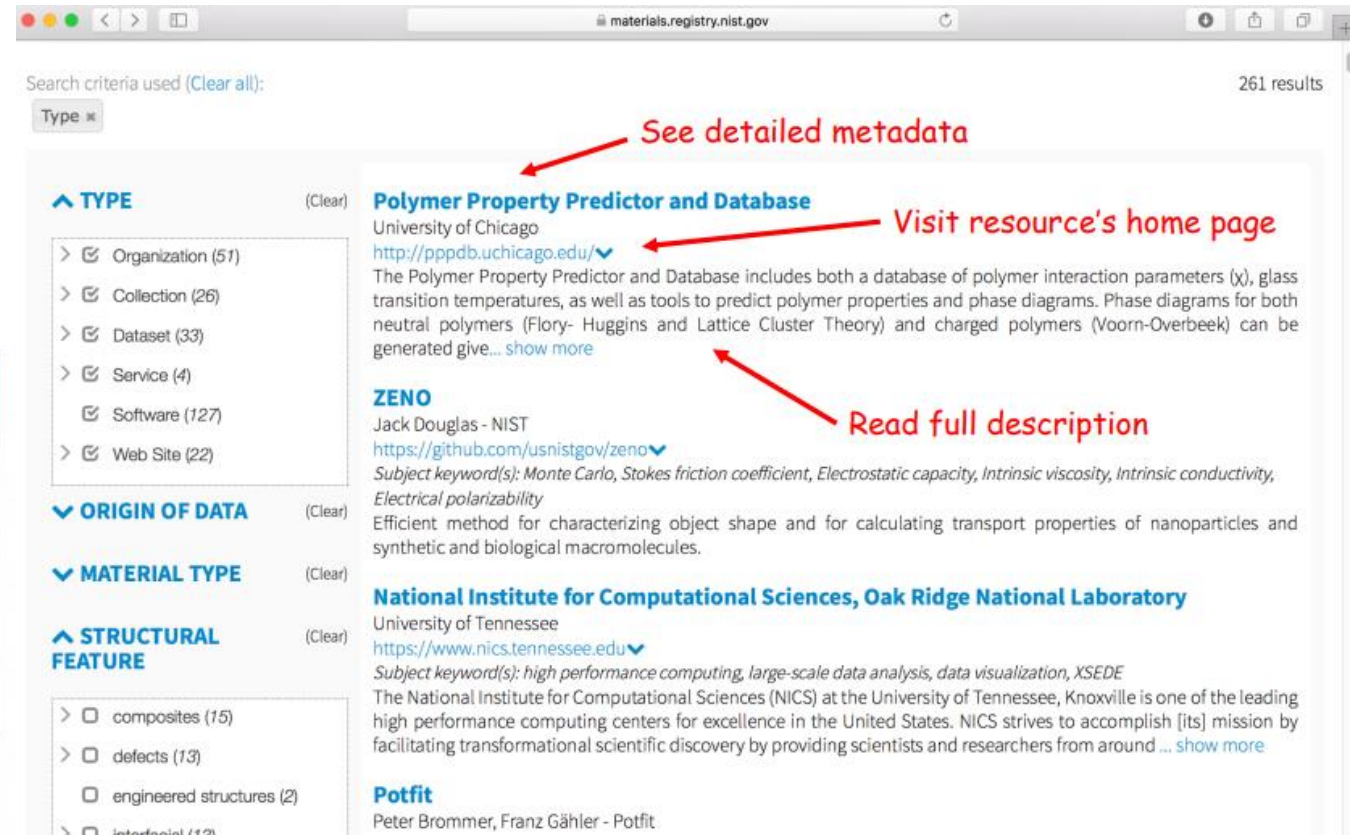
Add your own projects and data



Find Materials Data

This system allows for the registration of materials resources, bridging the gap between existing resources and the end users. The Materials Resource Registry functions as a centrally located service, making the registered information available for research to the materials community.

This is being developed at the National Institute of Standards and Technology and is made available to



ATOMISTIC SIMULATIONS FOR INDUSTRIAL NEEDS

2-3 day workshop in Rockville, MD in August, 2020

Talks and discussions fostering interactions between

- Potentials developers
- Tool developers
- Academic and industrial colleagues
- Machine learning and data analysis experts

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LINKS

<https://www.ctcms.nist.gov/potentials>
<https://www.ctcms.nist.gov/potentials/atomman>
<https://www.ctcms.nist.gov/potentials/iprPy>

NIST Interatomic Potentials Repository
atomman documentation
iprPy documentation

<https://materials.registry.nist.gov/>
<https://jarvis.nist.gov/>

NIST Materials Resource Registry
JARVIS

<https://github.com/usnistgov/potentials>
<https://github.com/usnistgov/atomman>
<https://github.com/usnistgov/iprPy>

NIST Interatomic Potentials Database
atomman source code
iprPy source code

potentials@nist.gov

Contact us