

# **Bond Order Potentials for AlCuH and C** available from LAMMPS site or us

**2015 LAMMPS Users' Workshop and Symposium**

**UNM Continuing Education Building, NM, August 5-7, 2015**

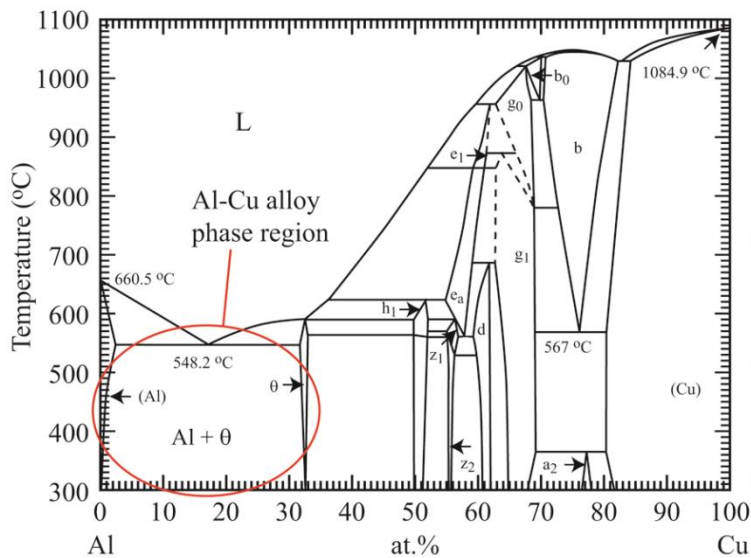
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**Sandia National Laboratories, USA**

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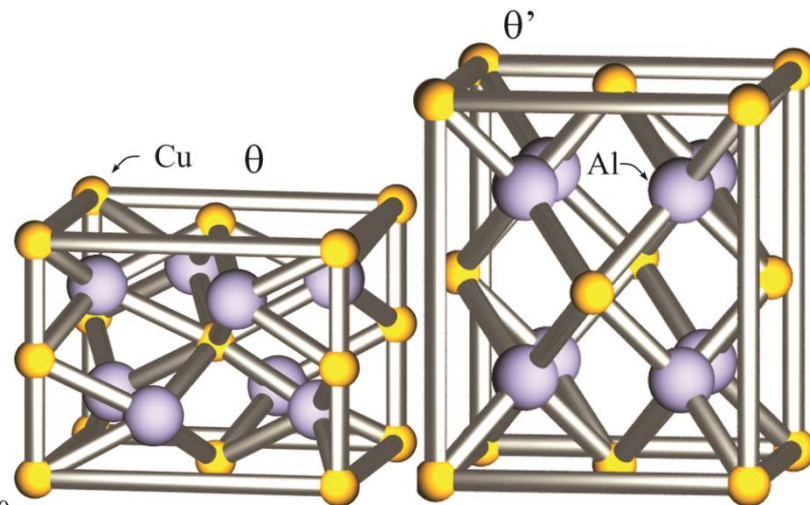
# Wish List for AlCuH Potential

1. A high stacking fault energy of Al observed in experiments;
2. Properties trends of a variety of stable and metastable structures;
3. Al-rich side of the Al-Cu phase diagram;
4. A reasonable positive heat of solution of Cu in Al;
5.  $H_2 \rightleftharpoons 2H$  chemical reaction;
6.  $Al_{1-x}H_x \rightarrow Al + H_2$  and  $Cu_{1-x}H_x \rightarrow Cu + H_2$  phase separations;
7. Robust MD simulations.

(a) Al-Cu phase diagram



(b) crystal structure of the  $\theta$  and  $\theta'$  phases



# Stacking Fault Energy of Al

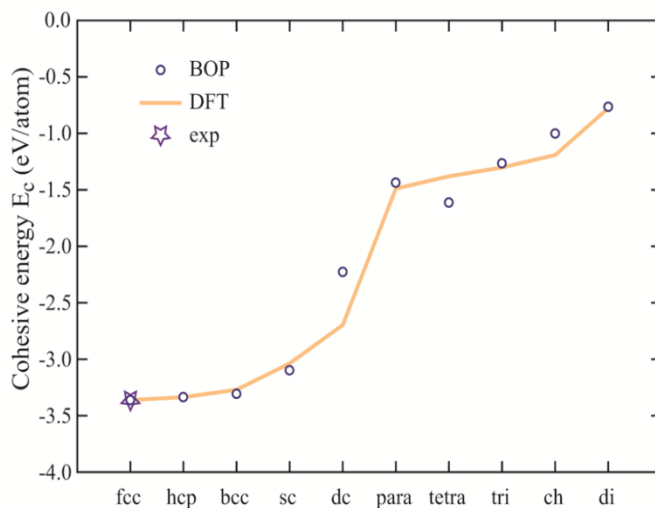
Model/Exp.	$\gamma_{100}$	$\gamma_{110}$	$\gamma_{111}$	$\gamma_{sf}$
EAM-CY	583	631	526	1
EAM-Mishin1	947	1013	873	141
EAM-BAM	1017	1154	1003	85
EAM-VC	862	969	829	71
EAM-MSAH	194	328	138	126
EAM-Zhou	868	958	832	44
EAM-MKBA	495	582	427	125
EAM-JNP	977	1055	910	0
MEAM	903	944	599	141
REAX-LJGS	481	483	427	0
REAX- Ojwang	810	848	711	1
BOP	979	1069	850	133
DFT [38]	1063	1098	987	-----
Exp. [59,60,61,62]	980-1140	980-1140	980-1140	120-144

**BOP captures a high stacking fault energy of Al.**

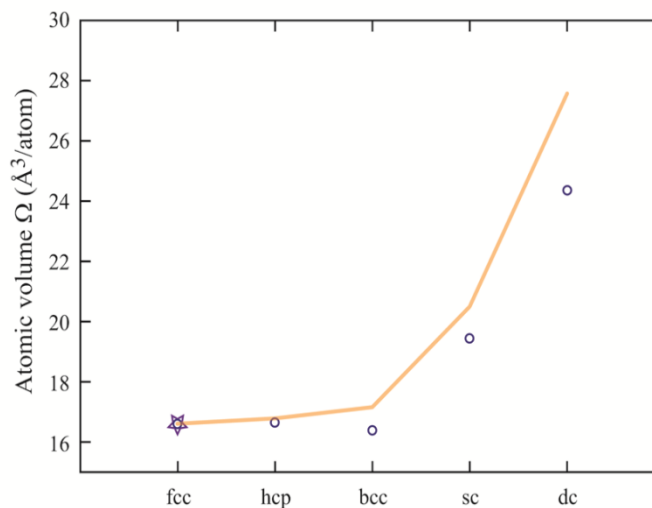
# Property Trends of Al and Cu

Al

(a) Cohesive energy  $E_c$

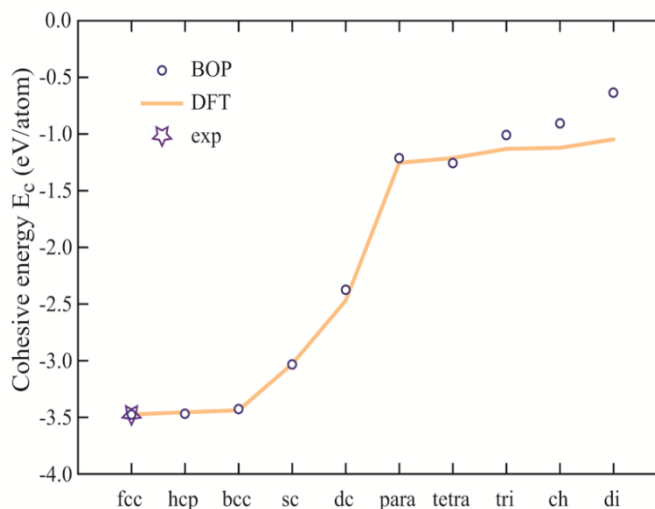


(b) Atomic volume  $\Omega$

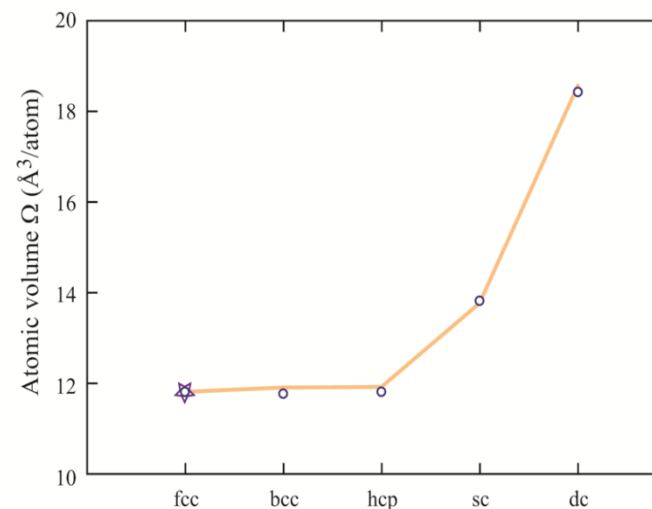


Cu

(a) Cohesive energy  $E_c$



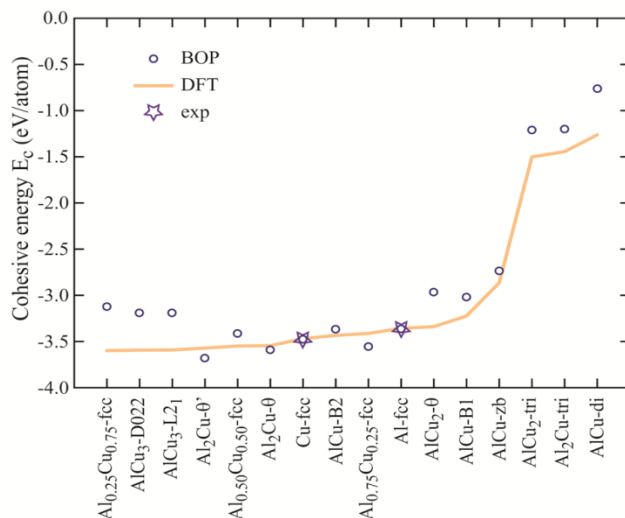
(b) Atomic volume  $\Omega$



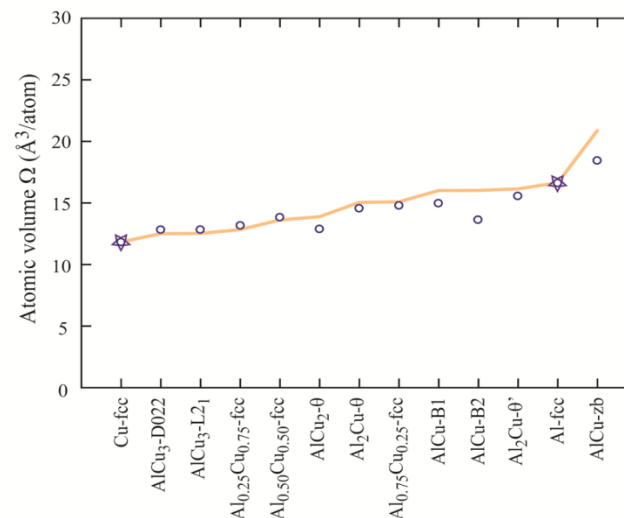
Pretty good property trends for Al and Cu.

# Property Trends of AlCu

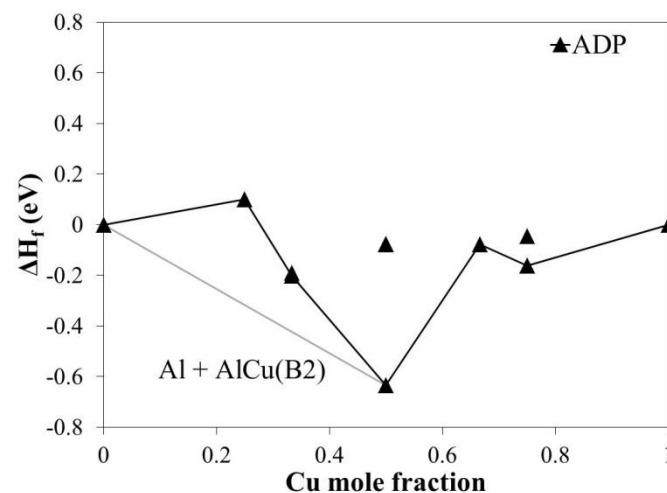
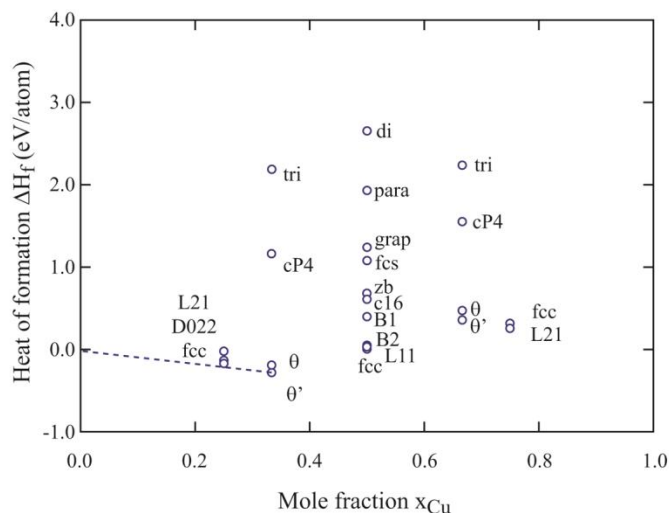
(a) Cohesive energy  $E_c$



(b) Atomic volume  $\Omega$



## Heat of Formation vs. Mole Fraction

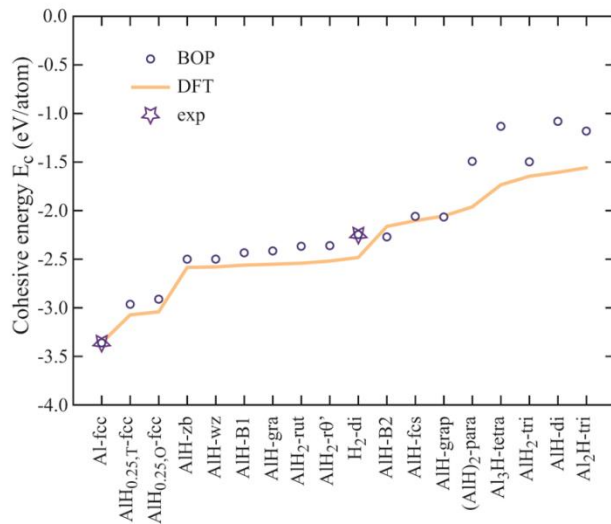


Pretty good property trends for AlCu.

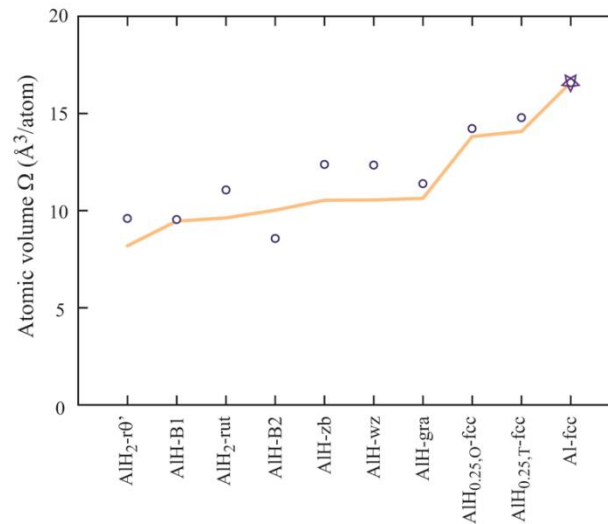
# Property Trends of AlH and CuH

AlH

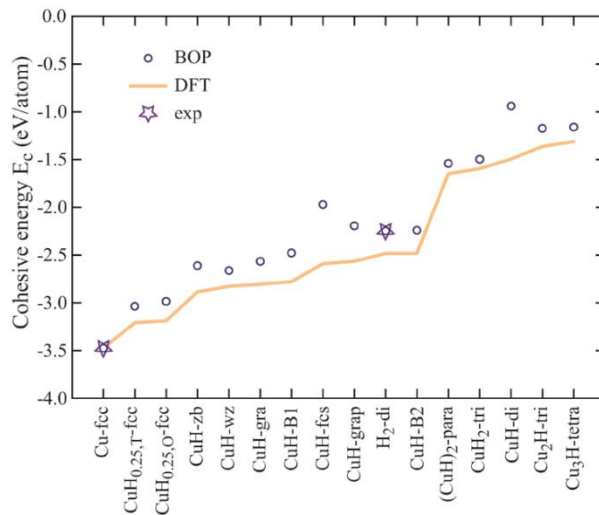
(a) Cohesive energy  $E_c$



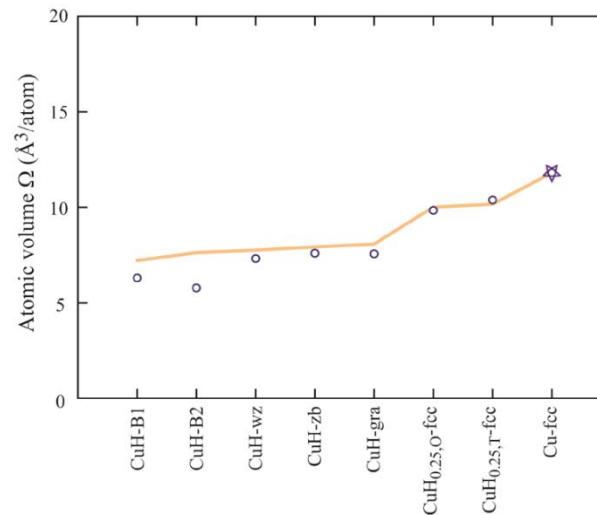
(b) Atomic volume  $\Omega$



(a) Cohesive energy  $E_c$



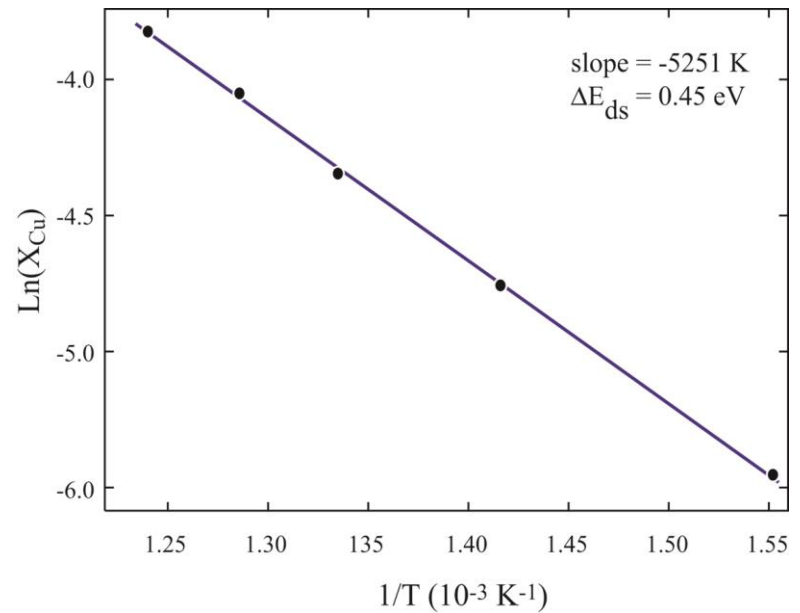
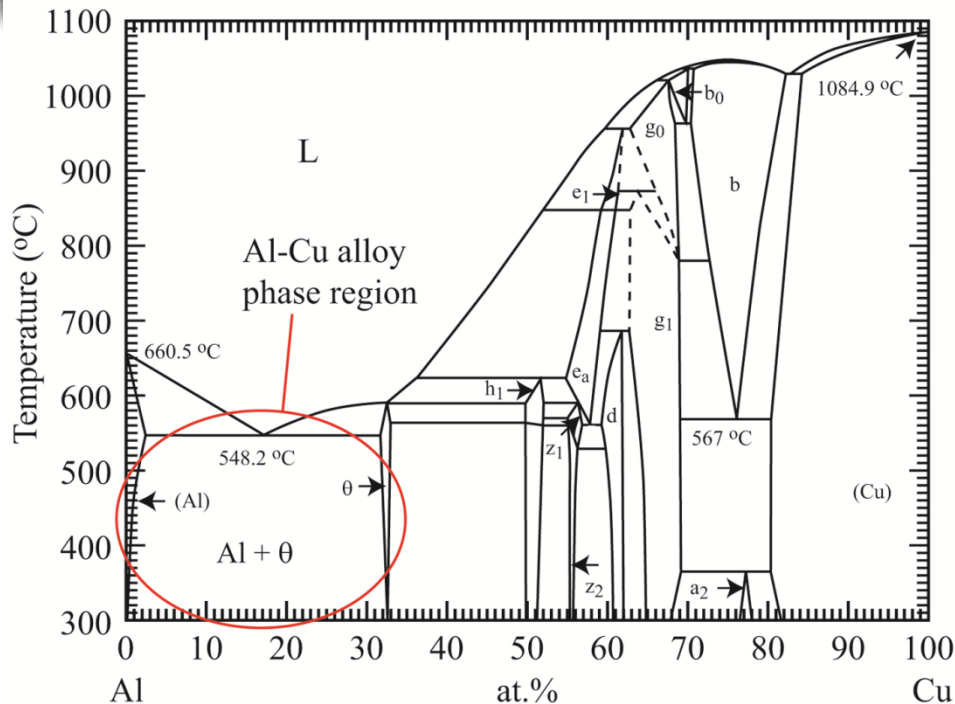
(b) Atomic volume  $\Omega$



CuH

Pretty good property trends for AlH and CuH.

# Dilute Heat of Solution of Cu in Al



## Results

1. Traditionally, heat of solution of Cu in Al is taken as energy change due to taking a Cu atom from Cu pool and putting it in Al pool;
2. Should really be the energy change due to taking a Cu atom from  $\text{Al}_2\text{Cu}$  pool and putting it in Al pool;
3. Must be a positive number

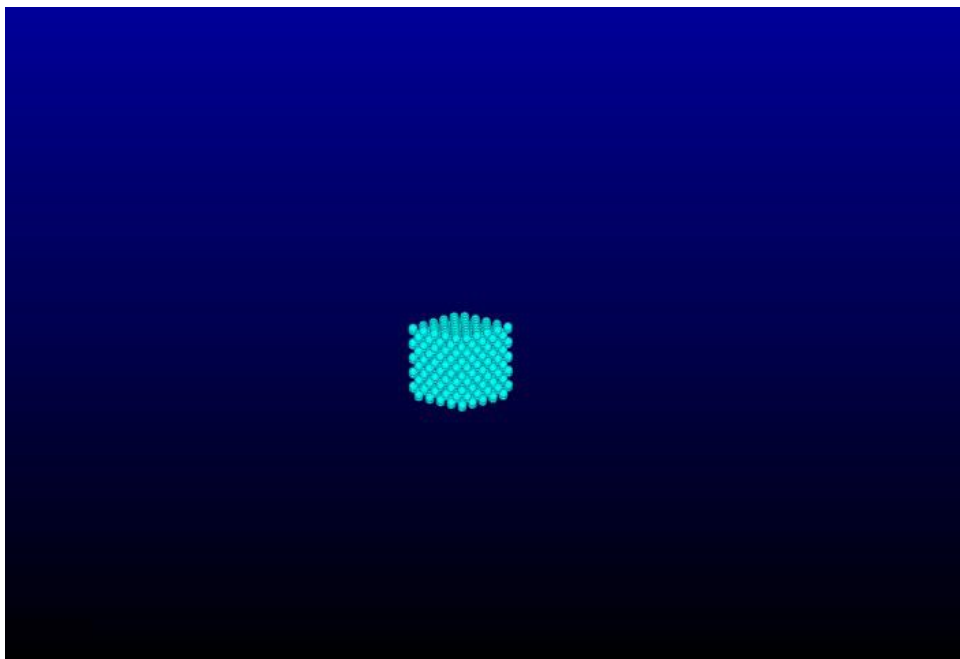
ADP	1.09
EAM-CY	-0.06
BOP	0.14
DFT	0.40
Exp.	0.45

X. W. Zhou, D. K. Ward, and M. E. Foster, manuscript to be submitted, potential available upon request

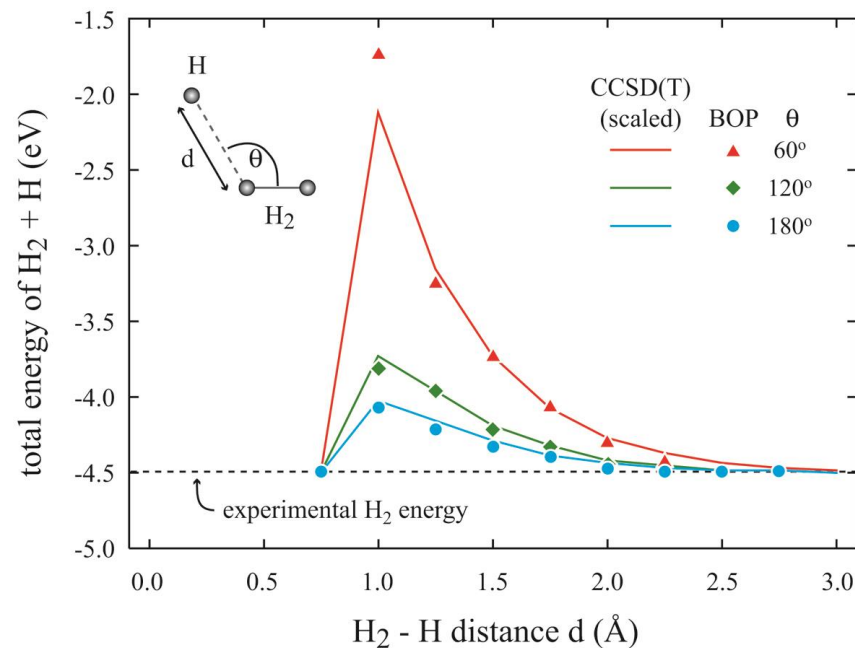


# $\text{H}_2 + \text{H} \rightarrow \text{H} + \text{H}_2$ Chemical Reaction

## Hydrogen crystal to $\text{H}_2$ gas



## $\text{H}_2 + \text{H} \rightarrow \text{H} + \text{H}_2$ energy profiles

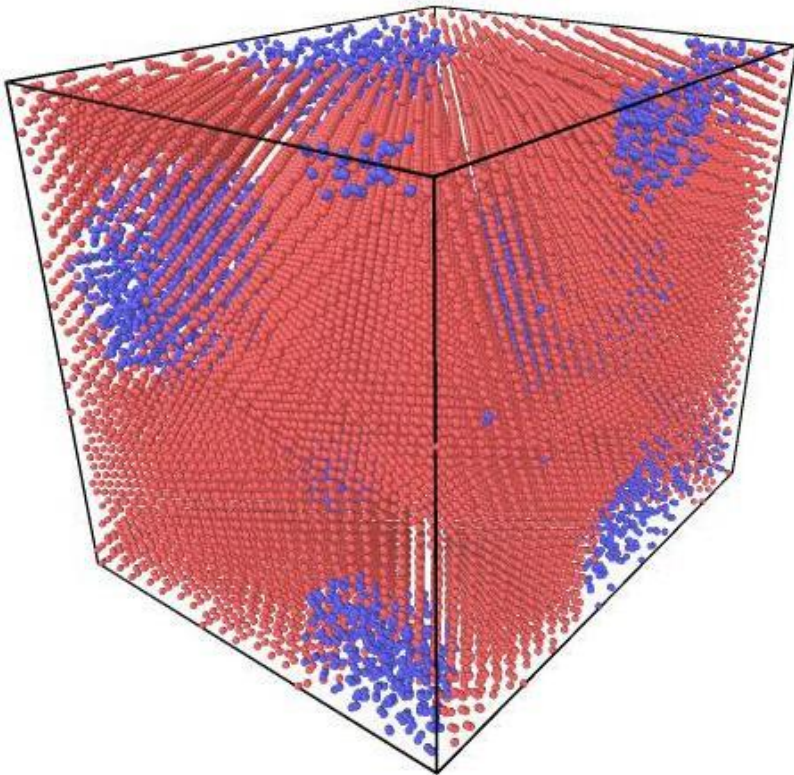


The BOP we developed has captured the  $\text{H}_2 + \text{H} \rightarrow \text{H} + \text{H}_2$  reaction, paper has been published:  
X. W. Zhou, D. K. Ward, M. Foster, J. A. Zimmerman, J. Mater. Sci., 50, 2859 (2015).

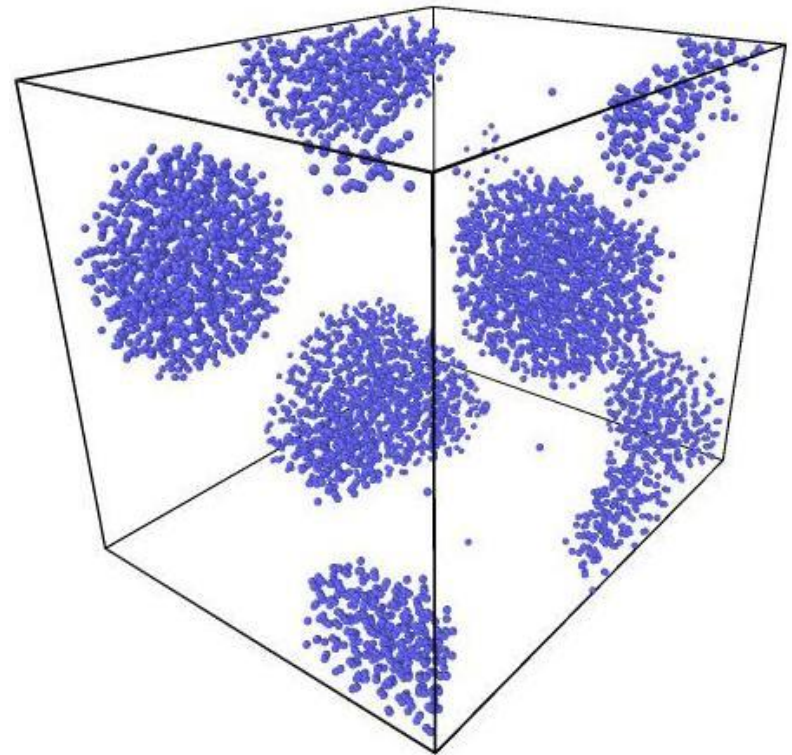


# Al + H<sub>2</sub> and Cu + H<sub>2</sub> Phase Separation

(a)  $N_{\text{H}}/N_{\text{Cu}} = 0.20$  with both  
Cu and H shown



(b) Only H shown

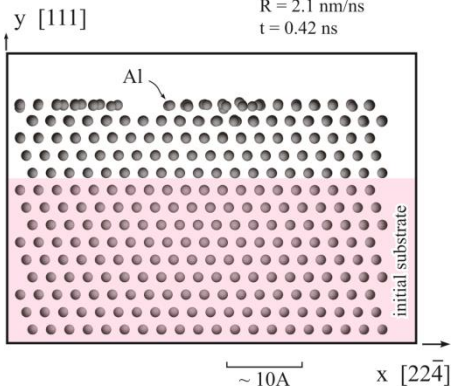


Phase separation is less pronounced in Al-H (see next slide), but Al forms AlH<sub>3</sub> compound.

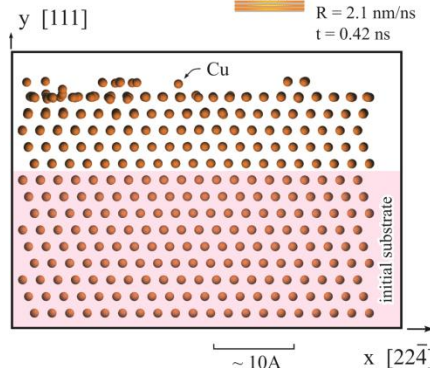
# Robust MD Simulations

## Al and Cu growth without H

$E_i = 1.0$  eV  
 $T = 300$  K  
 $R = 2.1$  nm/ns  
 $t = 0.42$  ns

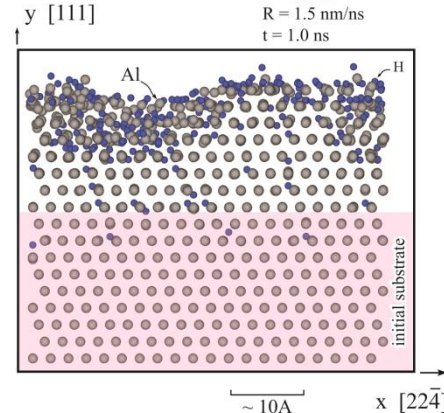


$E_i = 1.0$  eV  
 $T = 300$  K  
 $R = 2.1$  nm/ns  
 $t = 0.42$  ns

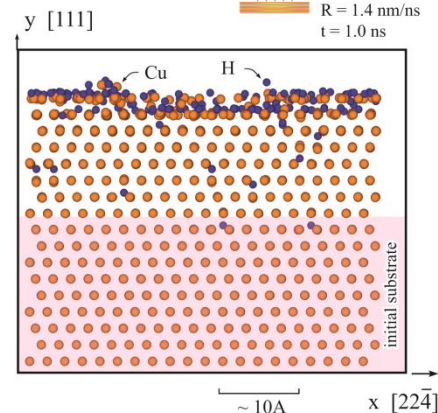


## Al and Cu growth with H

$E_i = 0.02$  eV  
 $\text{Al:H}_2 = 2:1$   
 $T = 600$  K  
 $R = 1.5$  nm/ns  
 $t = 1.0$  ns



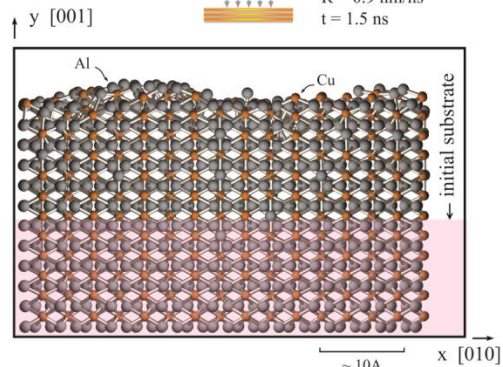
$E_i = 0.02$  eV  
 $\text{Cu:H}_2 = 2:1$   
 $T = 800$  K  
 $R = 1.4$  nm/ns  
 $t = 1.0$  ns



## $\text{Al}_2\text{Cu}$ ( $\theta$ and $\theta'$ ) growth without H

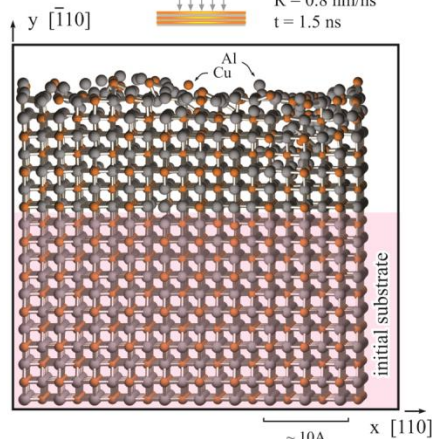
(a) Growth of  $\theta$

$E_i = 1.0$  eV  
 $\text{Al:Cu} = 2$   
 $T = 600$  K  
 $R = 0.9$  nm/ns  
 $t = 1.5$  ns



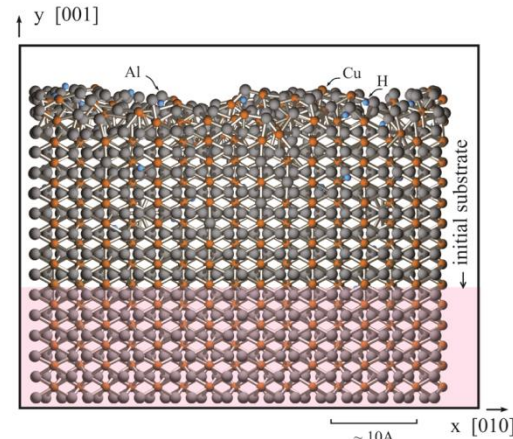
(b) Growth of  $\theta'$

$E_i = 1.0$  eV  
 $\text{Al:Cu} = 2$   
 $T = 500$  K  
 $R = 0.8$  nm/ns  
 $t = 1.5$  ns



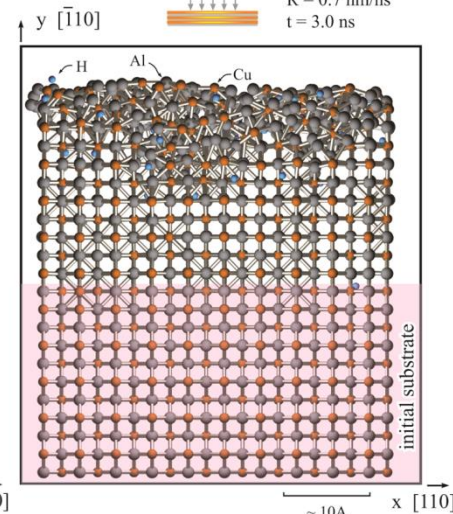
(a) Growth of  $\theta$

$E_i = 1.0$  eV  
 $\text{Al:Cu:H}_2 = 2:1:1$   
 $T = 600$  K  
 $R = 0.7$  nm/ns  
 $t = 3.0$  ns



(b) Growth of  $\theta'$

$E_i = 1.0$  eV  
 $\text{Al:Cu:H}_2 = 2:1:1$   
 $T = 500$  K  
 $R = 0.7$  nm/ns  
 $t = 3.0$  ns





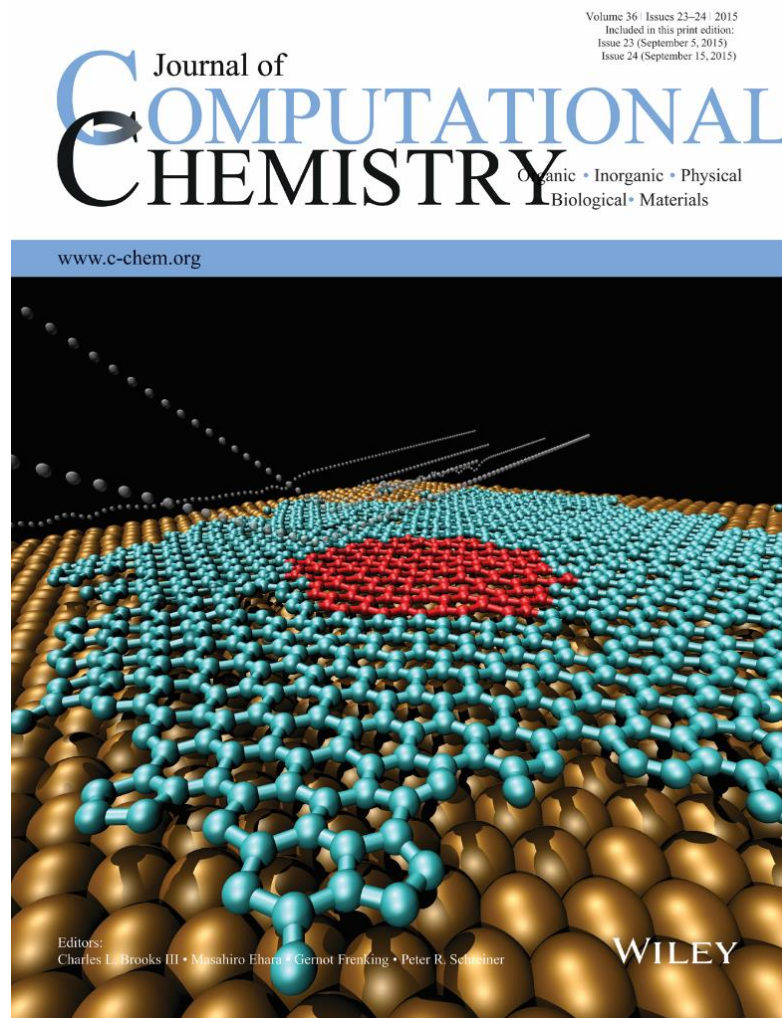
# Dream Wish for C Potential

1. Direct MD simulation of graphene;
2. Energy transferability between graphene, graphite, diamond;
3. Robust MD simulations.

Our goal is to complement the existing C potentials:

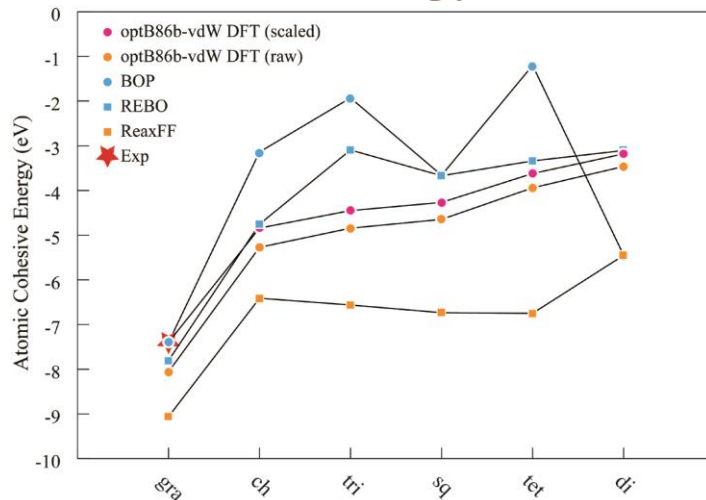
1. D. W. Brenner, O. A. Shenderova, J. A. Harrison, S. J. Stuart, B. Ni, and S. B. Sinnott, *J. Phys.: Condens. Matter*, 14, 783 (2002);
2. N. A. Marks, *Phys. Rev. B*, 63, 035401 (2001);
3. A. C. T. van Duin, S. Dasgupta, F. Lorant, and W. A. Goddard III, *J. Phys. Chem.*, 105, 9396 (2001).

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page 1719.

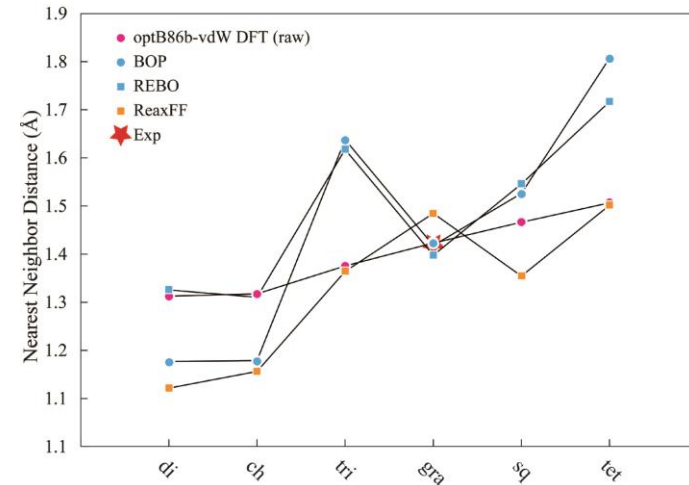


# Property Trends

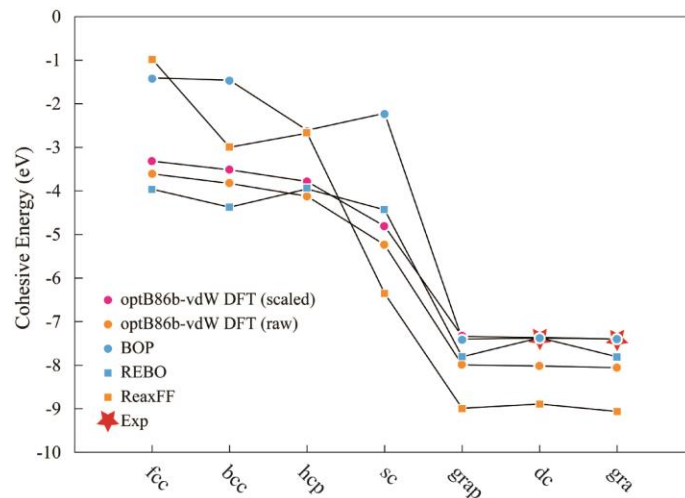
## Cluster Energy Trends



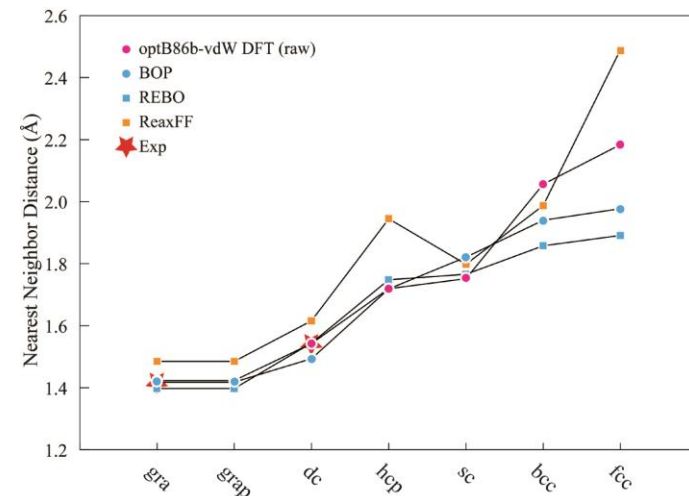
## Cluster Size Trends



## Lattice Energy Trends



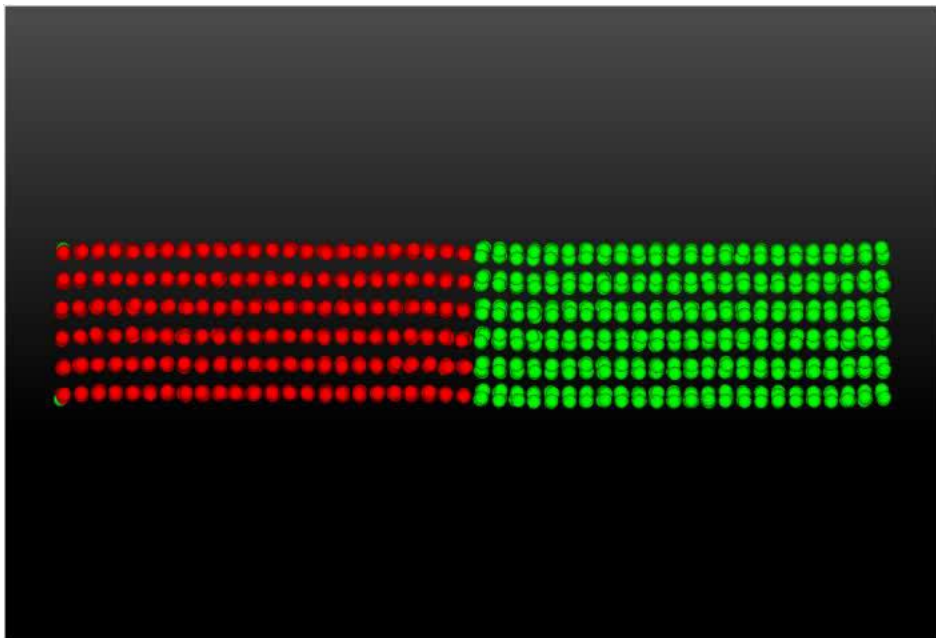
## Lattice Size Trends



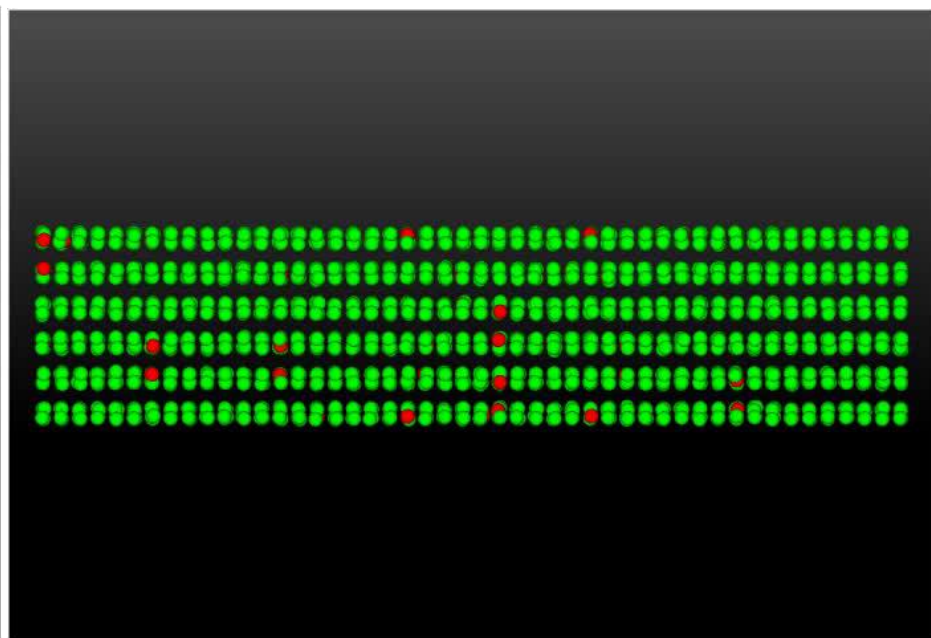
# Robust MD Simulations

## Graphite $\Leftrightarrow$ Diamond Transformation

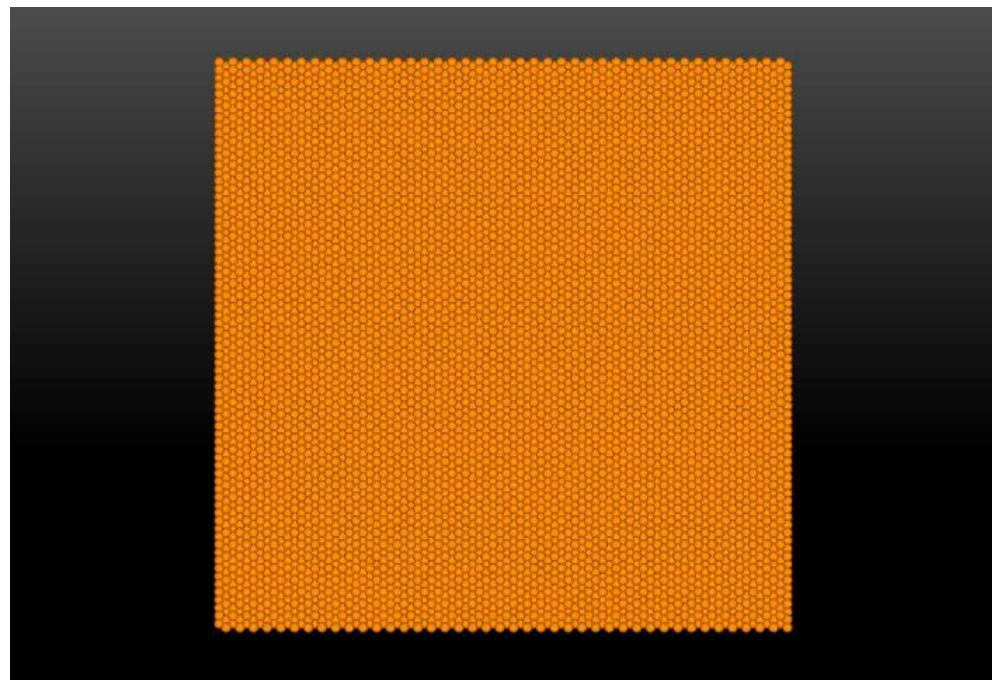
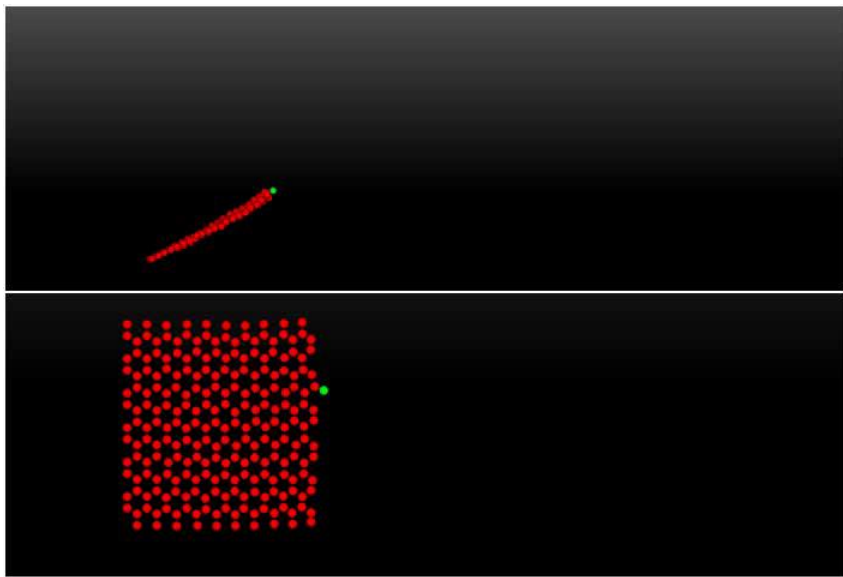
$T = 2000 \text{ K}$ ,  $P = 0.6 \text{ Mbars}$



$T = 1600 \text{ K}$ ,  $P = -0.6 \text{ Mbars (tensile)}$

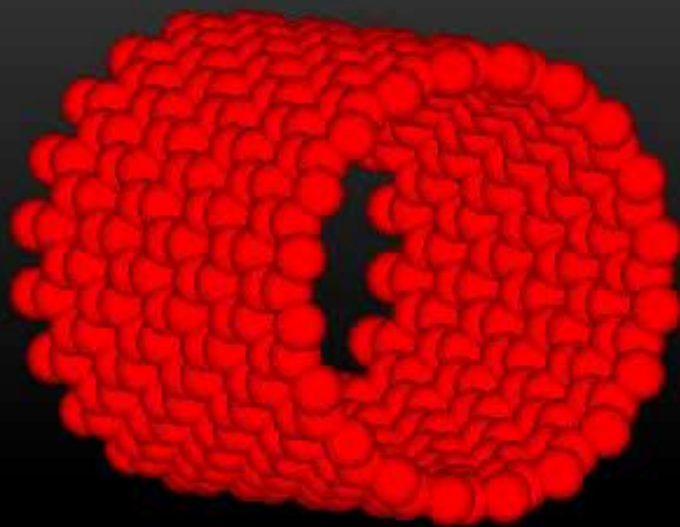


# Robust MD Simulations: Graphene Growth



# Robust MD Simulations

## Nanotube Growth





# Conclusions

1. Two bop potentials, AlCuH.bop, and C.bop, have been released to lammps package;
2. The AlCuH.bop meets the six+ criteria needed to study mechanical properties of AlCu alloys;
3. C.bop enables MD simulation of graphene growth.