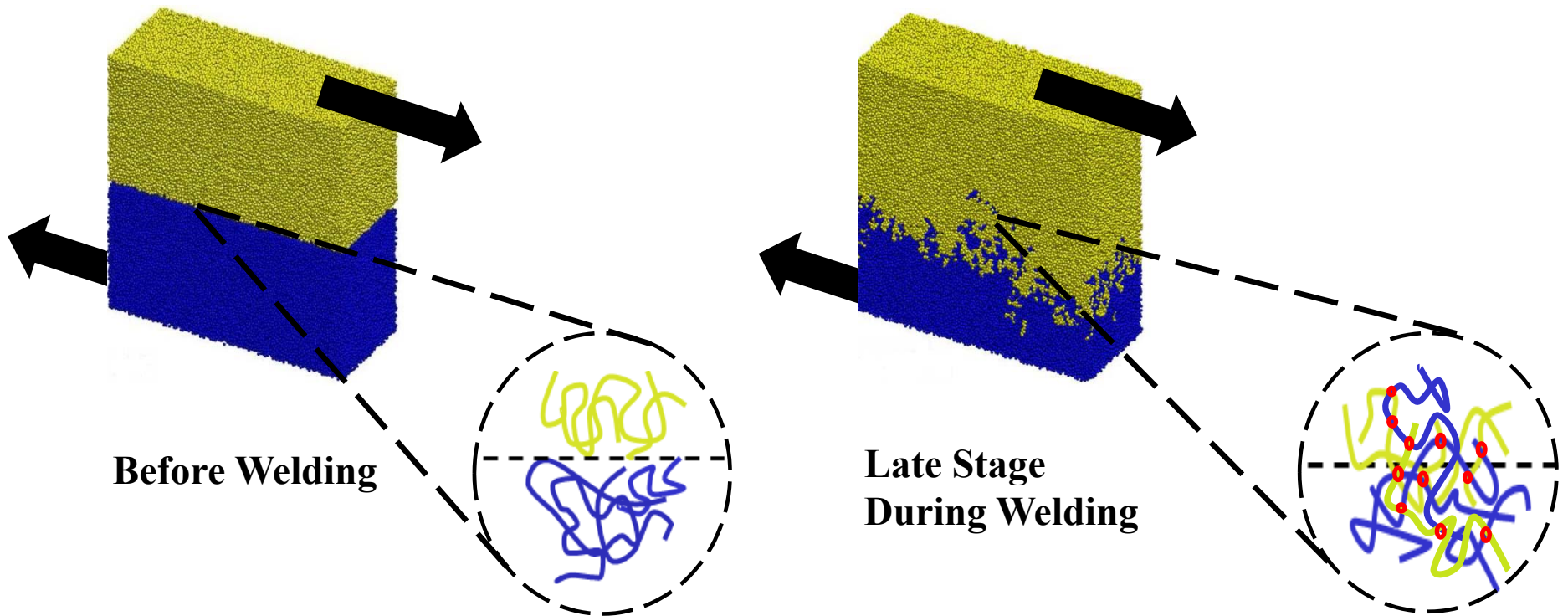


# Shear Failure of Polymer Welds and Entanglements

*How does interfacial strength develop during welding?  
How does the development correlate with  
the evolution of interfacial structure?*

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# MD Simulation and PPA algorithm

## Nonequilibrium molecular dynamics (MD) simulation

- Lennard-Jones potential between all spherical beads

Breakable bond potential

**ratio of the forces at which the covalent and van der Waals bonds break  $\sim 100$**

- Simple shear

**constant strain rate  $d\gamma_{xz}/dt = 2 \times 10^{-4}$**

Periodic boundary conditions within the shear plane

Temperature control

## Primitive Path Analysis (PPA)

- Fix chain ends

Deactivate intrachain excluded-volume interactions

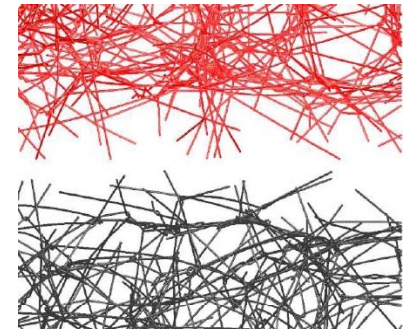
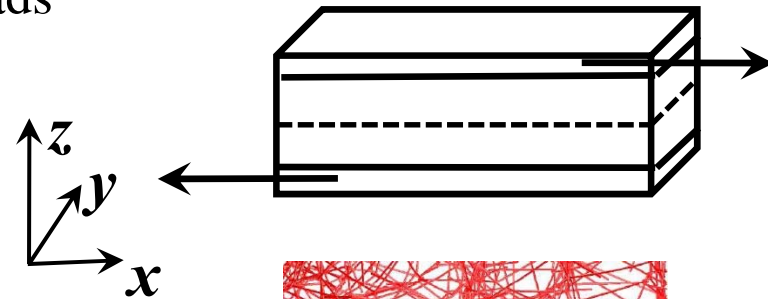
Retain interchain excluded-volume interactions

- **Minimize energy by cooling the system down to  $T \sim 0$**

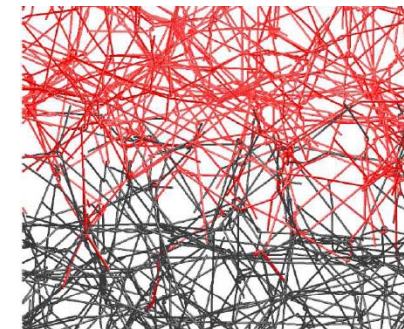
- Bond forces try to reduce the bond length to zero and pull chains taut

- Insert extra beads to reduce the effects due to chain thickness

- Contacts of Primitive Paths  $\longleftrightarrow$  Entanglements



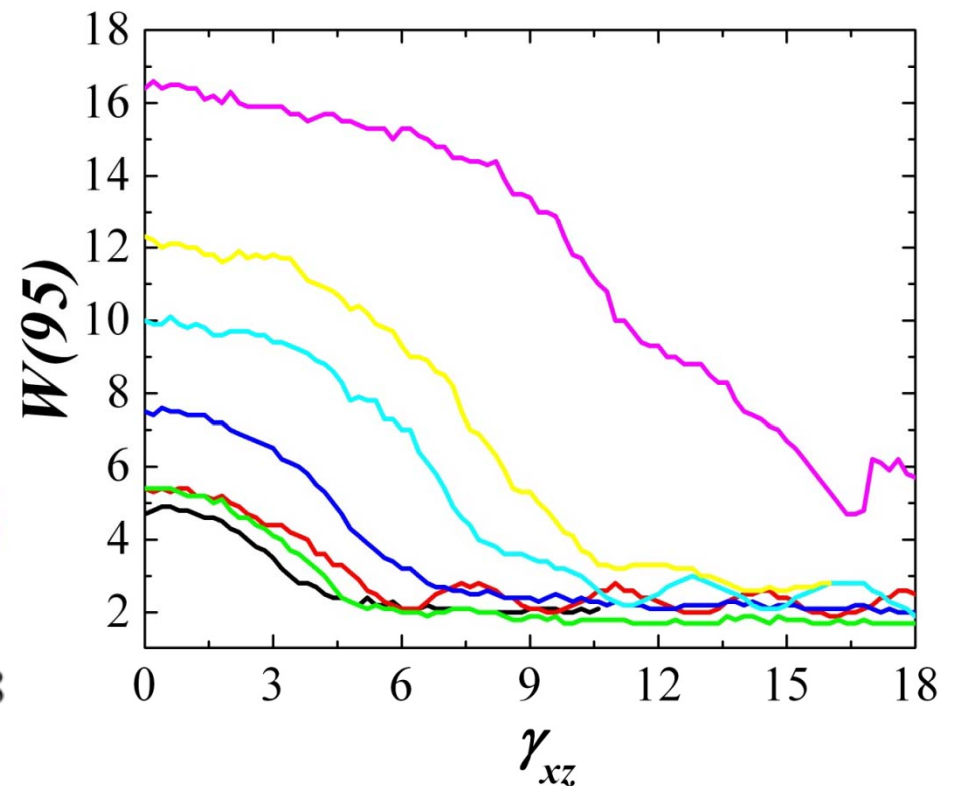
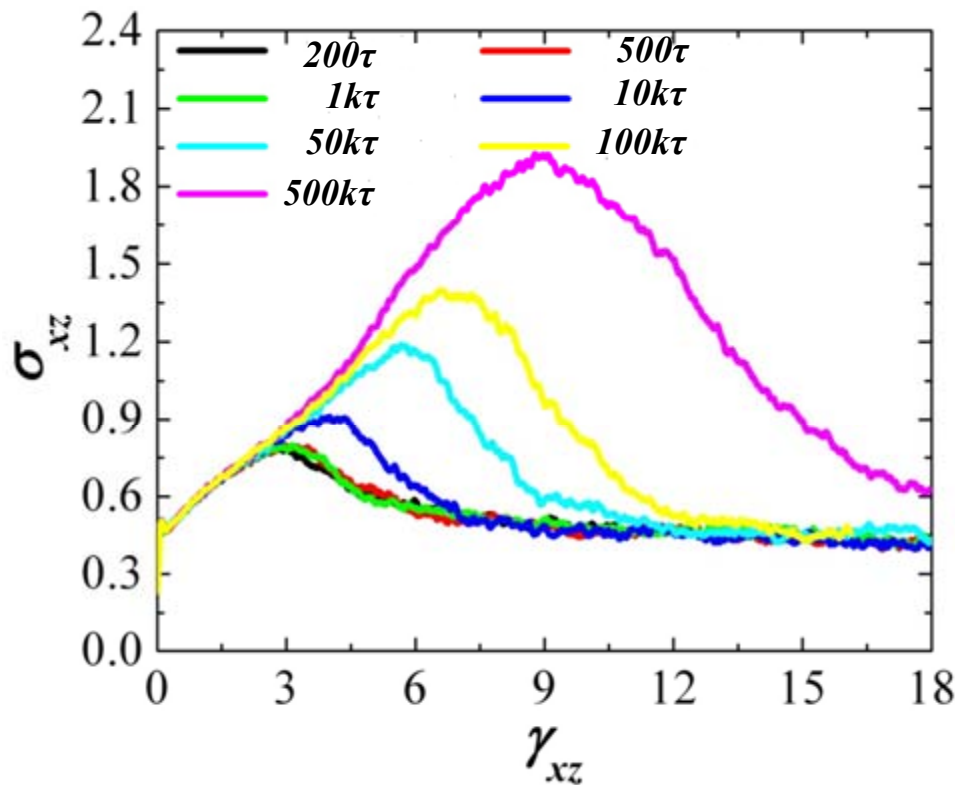
*PPs before welding*



*PPs at late stage of welding*

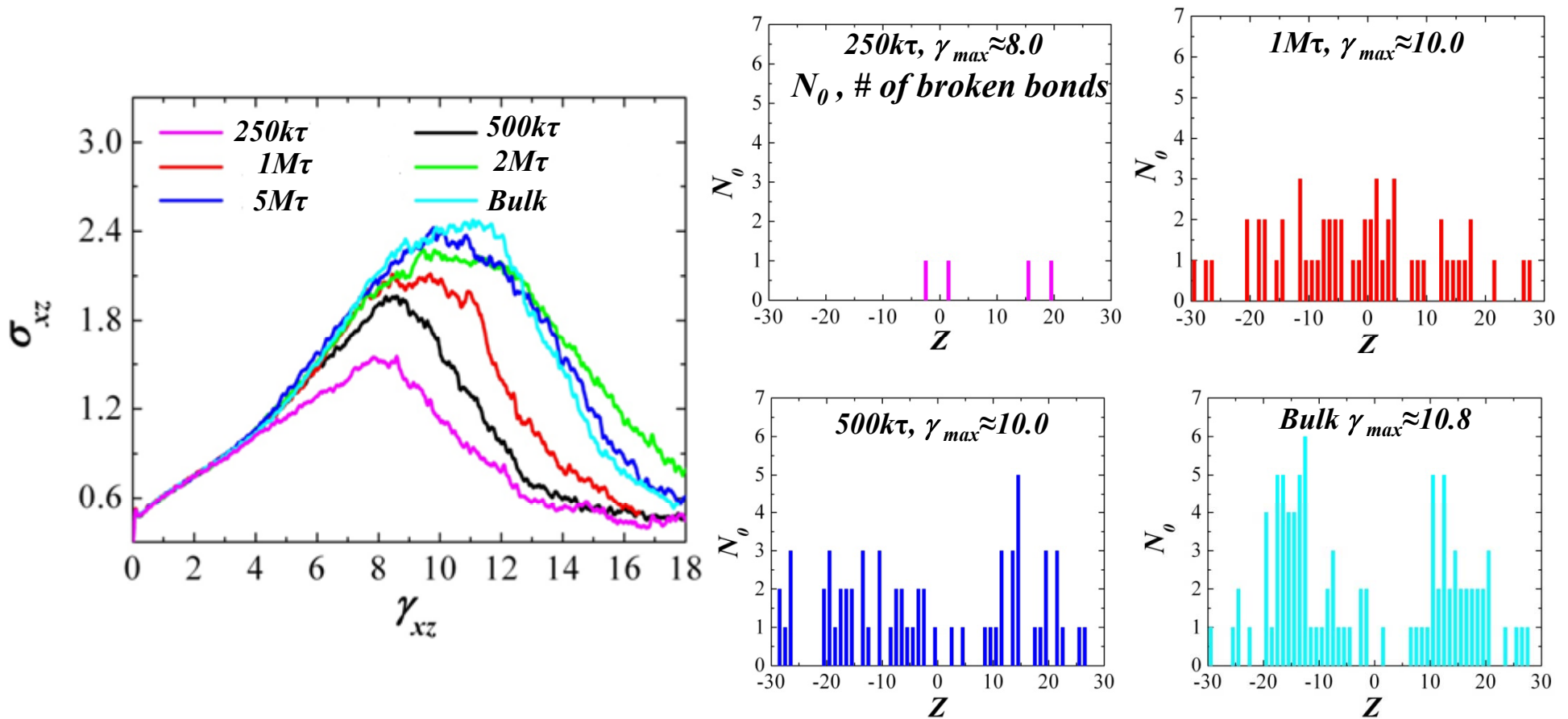
# Interface fails by pure chain pullout at small $t$

- Interfacial width decreases with shear strain  
*W(95), the distance between the planes where 95% atoms belong respectively to original upper and lower parts*
- At steady state,  $\sigma_{xz}$  is uniform along  $Z$   
**Strain hardening in the entangled regions away from the interface**  
**→ Strain localization near the interface**
- Universal final interfacial width and shear stress



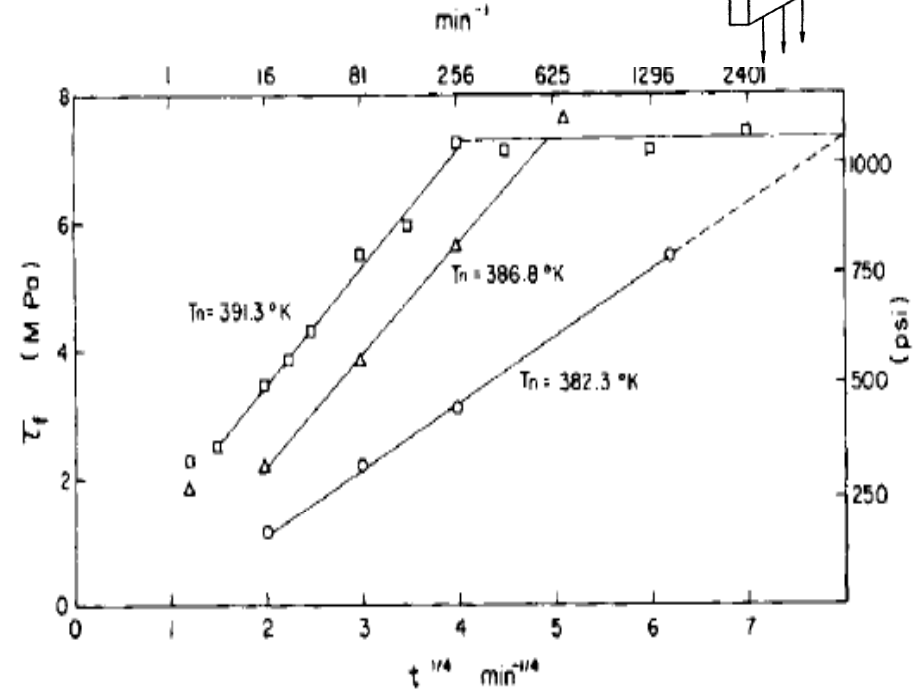
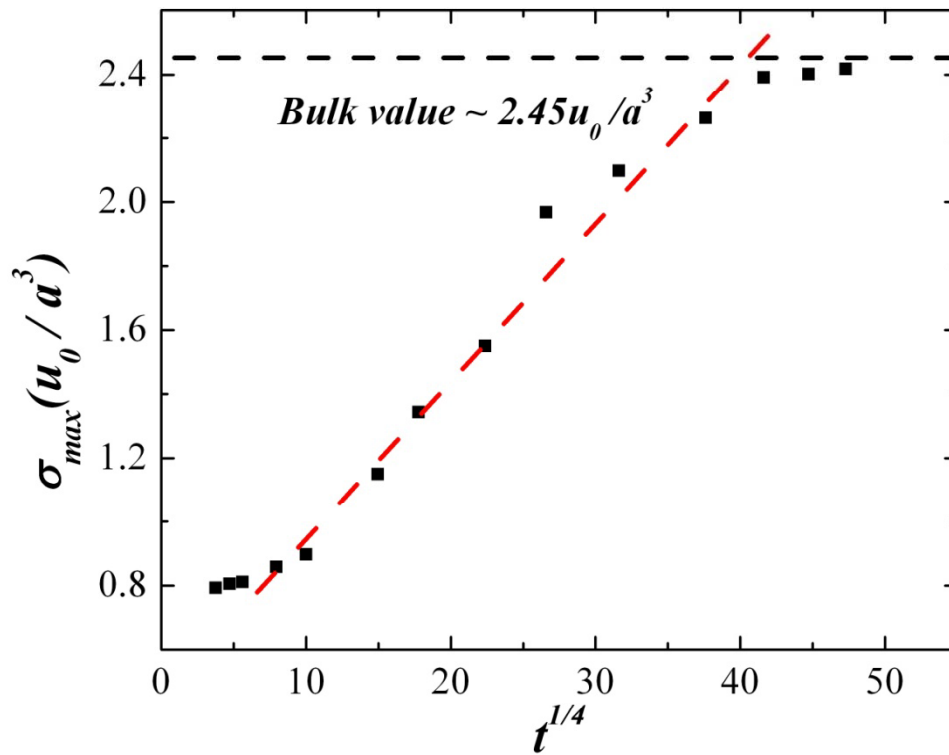
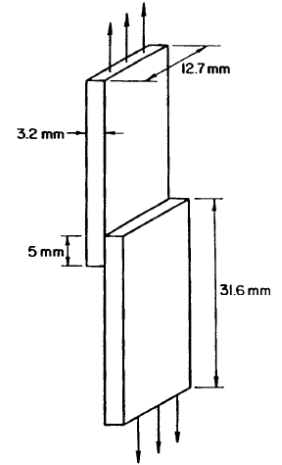
# Interface fails by chain scission at large $t$ and in bulk states

- Broken bonds show up at large  $\gamma$   
*Not only near the interface*
- After chain scission sets in, the stress-strain behavior starts to saturate towards the bulk result



# Time Dependence of the Maximum Shear Stress before Failure

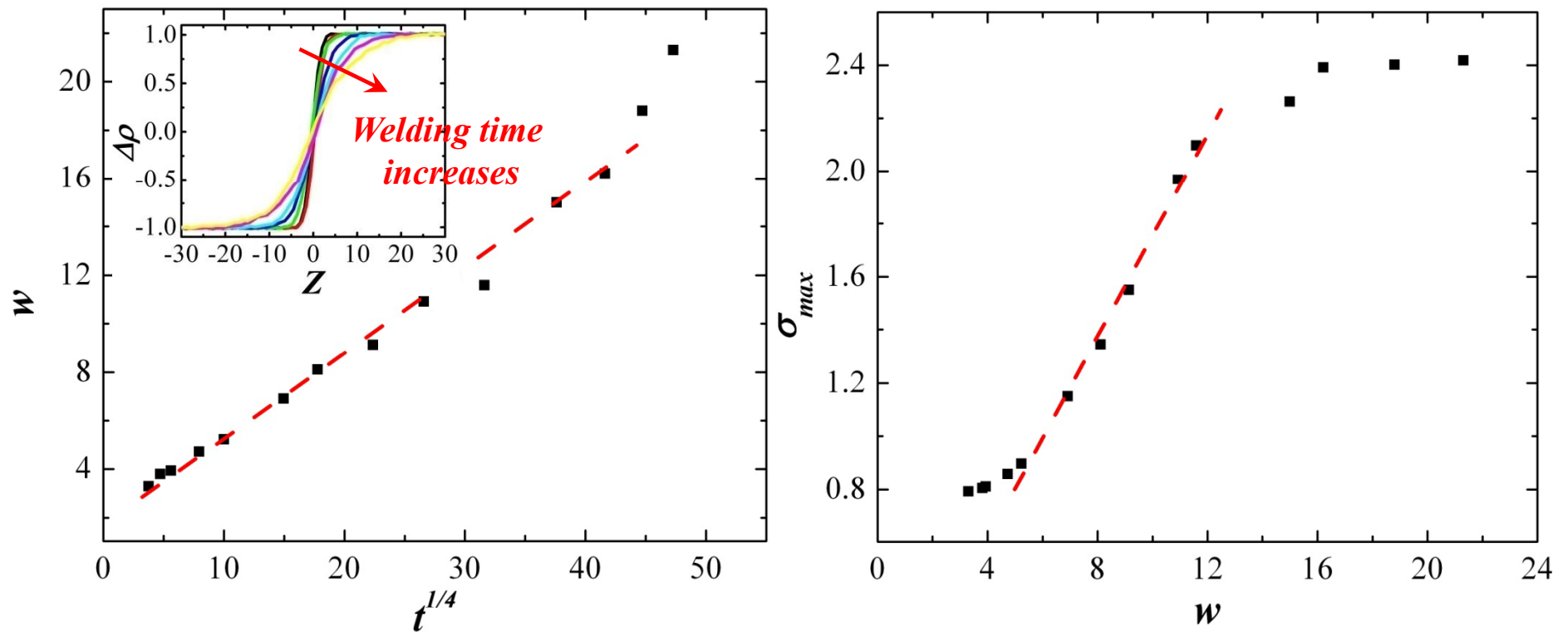
- $t^{1/4}$  scaling law in the intermediate regime
- After chain scission sets in,  $\sigma_{max}$  saturates towards the bulk value
- Agree with the experimental results by the *Lap-Shear Joint Method*



*D.B.Kline & R.P.Wool (1988)*

# $\sigma_{max}$ Correlates with Interfacial Width before Saturation

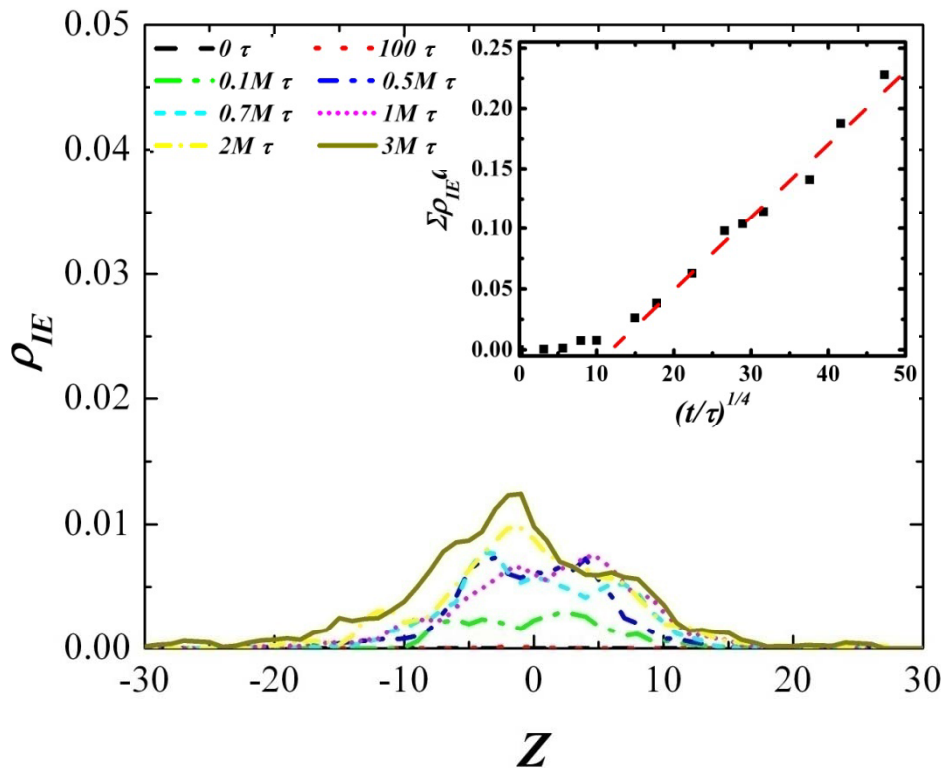
- $t^{1/4}$  scaling law of the interfacial width  $w$   
 $w$  is obtained by fitting the initial density difference profiles using  $\tanh(2z/w)$   
**Consistent with theoretical predictions based on reptation dynamics**
- Before saturation,  $\sigma_{max}$  correlates well with  $w$



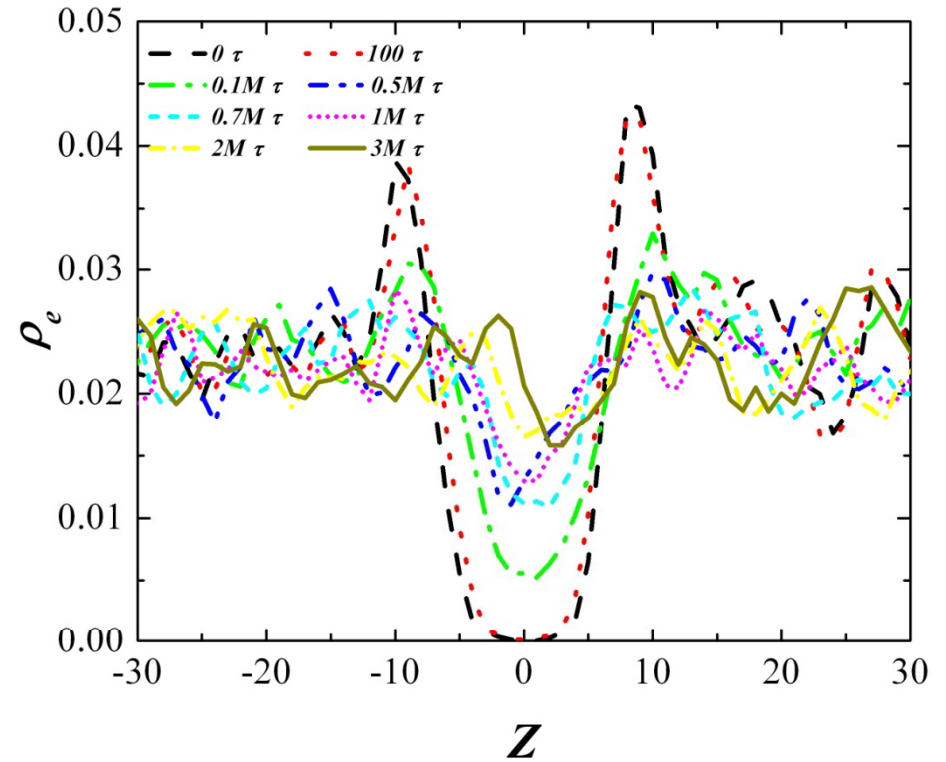
# $\sigma_{max}$ Correlates with Integrated Interfacial Entanglement Density Saturates as Entanglement Density Profile Approaches the Bulk Result

- Interfacial Entanglements (IEs) form between chains from the opposite sides  
 $\Sigma\rho_{IE}$  obeys  $t^{1/4}$  scaling law
- Bulk shear strength is fully recovered when entanglement density  $\rho_e$  equals its bulk value

*Entanglements between chains  
from opposite sides*



*Entanglements between chains  
from any side*



# Conclusions

## *Power law dependence of interfacial shear strength on welding time*

- $\sigma_{max}$  rises with welding time as  $t^{1/4}$ , in agreement with experiment and theory of reptation dynamics.
- At small  $t$ , the interface fails via pullout of chain ends.  
At large  $t$  and for the bulk, shear failure is through chain scissions.

## *Correlation of interfacial strength with interfacial structure*

- $\sigma_{max}$  correlates with the interfacial width  $w$  and  $\Sigma\rho_{IE}$  and both increase as  $t^{1/4}$ .
- The crossover to bulk strength coincides with the evolution of  $\rho_e$  to its bulk distribution.